Structures of Local Rearrangements in Soft Colloidal Glasses

Xiunan Yang,¹ Rui Liu,¹ Mingcheng Yang,¹ Wei-Hua Wang,^{2,*} and Ke Chen^{1,†}

¹Beijing National Laboratory for Condensed Matter Physics and Key Laboratory of Soft Matter Physics, Institute of Physics,

²Institute of Physics, Chinese Academy of Sciences, Beijing 100190, People's Republic of China

(Received 6 November 2015; revised manuscript received 22 January 2016; published 9 June 2016)

We image local structural rearrangements in soft colloidal glasses under small periodic perturbations induced by thermal cycling. Local structural entropy S_2 positively correlates with observed rearrangements in colloidal glasses. The high S_2 values of the rearranging clusters in glasses indicate that fragile regions in glasses are structurally less correlated, similar to structural defects in crystalline solids. Slow-evolving high S_2 spots are capable of predicting local rearrangements long before the relaxations occur, while fluctuationcreated high S_2 spots best correlate with local deformations right before the rearrangement events. Local free volumes are also found to correlate with particle rearrangements at extreme values, although the ability to identify relaxation sites is substantially lower than S_2 . Our experiments provide an efficient structural identifier for the fragile regions in glasses and highlight the important role of structural correlations in the physics of glasses.

DOI: 10.1103/PhysRevLett.116.238003

Understanding the structure-property connections in glasses is one of the most challenging problems in condensed matter physics. The lack of a clear characterization of the structures in glasses has hindered the formulation of general theories for glass transition and deformation of glasses [1–7]. The most elementary relaxation events in glasses are the local atomic rearrangements that play similar roles in the mechanical properties of glasses as the structural defects in crystals. Local regions prone to rearrangements in glasses are often known as flow units [8,9], shear transformation zones [10,11], soft spots [12–15], or geometrically unfavored motifs [16]. Unlike crystalline solids, whose defects are easily identified from a periodic lattice, no distinguishing structures have been found for the fragile regions of glasses in an apparently disordered background. Without a clear structural indicator for fragile regions in glasses, many studies rely on phenomenological models such as soft glassy rheology to understand macroscopic properties of amorphous materials [17].

Early studies attempted to connect a particle's propensity to rearrange to its immediate environment. Spaepen proposed a free-volume theory that favors sites with large free volumes for local rearrangements in hard-sphere glasses [18]. For soft-sphere systems, which include metallic glasses, Egami and co-workers suggest that regions with extreme local stresses, which correspond to particles with either extremely large or extremely small free volumes, are potential "defects" sites [19,20]. Direct experimental examinations of these scenarios, however, are rare due to the difficulty of measuring local free volumes or atomic stresses in glassy materials [21]. More recently, machine learning methods are employed to search through multidimensional parameter space and use a combination of structural features to identify fragile regions in disordered solids [22,23].

Phonon modes, which reflect the collective excitations in glasses, are shown to be a powerful identifier of fragile regions in glasses. Soft spots defined by the low-frequency quasilocalized soft modes are shown to overlap significantly with rearranging regions in glasses [12-14]. However, measurements of the local geometric structures of soft spots did not find any distinctive structural characteristic for these regions [14]. Recent theories and experiments suggest that, in glasses, there may exist amorphously correlated structures formed during the glass transition [24–29], and the dynamics-structure correlation goes beyond single-particle measures [30]. Therefore, the search for "defects" in glasses needs to probe correlations (or lack thereof) on length scales greater than the first-neighbor shell. Recent simulations by Tong and Xu [15] find strong correlations between the distributions of soft modes in glasses and structural entropy S_2 , raising the possibility that this correlation-based structural parameter may also be able to predict elementary deformations in glasses.

In this Letter, we employ video microscopy to directly image local deformations in quasi-2D colloidal glasses consisting of thermosensitive microgel particles and measure the structures of local rearranging regions under cyclic thermal perturbations. Thermal cycling tunes the sizes of soft colloidal particles and generates uniform compression or dilation when the particles swell or shrink. Similar to mechanical shearing [31–35], rearrangements during thermal cyclings are the results of local stress imbalances. But perturbations from thermal cycling impose no directional bias or boundary effects and allow the simultaneous activation of defects with different orientational preferences.

Chinese Academy of Sciences, Beijing 100190, People's Republic of China

We find that local free volumes are weakly correlated with particle rearrangements; particles with either extremely high or low free volumes are equally likely to rearrange. Local structural entropy S_2 derived from local pair correlation functions strongly correlates with observed local rearrangements. Rearranging clusters emerge and evolve with regions of high S_2 values, which are structurally less correlated or more disordered compared to the more stable background, similar to the topological defects in polycrystalline solids. The results presented here are for deeply jammed samples under small periodic perturbations. We have also observed qualitatively the same correlations in samples of different packing fractions and under different perturbation patterns [36].

The samples are prepared by loading a binary mixture of poly-N-isopropylacrylamide (PNIPAM) particles between two coverslips. The colloidal suspension spreads under capillary forces and forms a dense monolayer between the glass plates. The samples are then hermetically sealed using optical glue (Norland 63). A binary mixture is used to frustrate crystallization. The diameters of the particles are measured to be 1 and 1.3 μ m at 22 °C by dynamical light scattering, and the number ratio between large and small particles is close to 1. PNIPAM colloidal particles are temperature-sensitive particles whose diameters decrease when the temperature increases [37–40]. The interactions between PNIPAM particles are generally characterized to be repulsive soft-sphere with a hard core [41]. Rearrangements are induced by locally changing the sample temperature. The colloidal suspension is mixed with a small amount of nonfluorescent dye (Chromatch-Chromatint black 2232 liquid, 0.2% by volume). When illuminated by a mercury lamp, the dye absorbs the incident light and heats up a small area much larger than the field of view, while the rest of the sample remains at ambient temperature [42,43]. When the mercury lamp is turned off, the heated region rapidly recovers to its original temperature. The temperature increase from optical heating is calibrated to be about 0.2 K, and the sample reaches new thermal equilibrium in less than 1 s [36]. For this small temperature change, the samples remain in jammed glassy states at both temperatures, with a brief transient period in between.

The samples are aged on the microscope stage for 200 min before being periodically heated and cooled for a total of ten cycles. Each cycle lasts for 30 min, with 15 min each at the elevated and ambient temperature [36]. The samples are continuously imaged using standard bright-field microscopy at 20 fps for the duration of the experiment. There are ~3700 particles within the image frame, with a packing fraction of 0.88 at the ambient temperature. The trajectory of each particle is extracted by particle tracking techniques [44]. We limit our analyses on particles that are at least three diameters away from the image boundaries, which leaves ~2990 particles in a reduced field of view. Information of particles outside of this reduced field of view is utilized only for the calculation of parameters for particles within. Most particle rearrangements are observed within 60 s of a temperature switch. A video of one local rearranging event can be found in Supplemental Material [36].

We extract plastically rearranged regions by comparing particle positions at the last minute of each cycle with the positions of the particles 1 min before the same cycle; at both points, the sample is at thermal equilibrium with the ambient temperature. Separation cutoffs defined by the first minima on the particle pair correlation functions between different species are used to determine neighbor changes during thermal cycling [12,43]. Distributions of particle separations over time are also considered to avoid accidental misidentification [36]. Particles that changed neighbors during one cycle are grouped into clusters based on nearest-neighbor pairings, and clusters containing fewer than five particles are ignored [13,14]. Most neighbor changes are permanent; only about 2% of rearranging particles regain their lost neighbors in subsequent thermal cycles.

Figure 1 shows all rearranging clusters observed during the ten thermal cycles. Clusters from different cycles are shown in different colors. For a single cycle, the spatial distribution of rearranging clusters appears uncorrelated, with cluster separations much larger than typical cluster sizes. However, when the distributions of rearranging clusters from different thermal cycles are compared, it is clear that the clusters are concentrated in certain regions of the sample, as indicated by the large blue dotted circles in Fig. 1, instead of evenly scattered throughout the sample. Overlapping between neighboring clusters is rare, with only four overlapping particles for the duration of the experiment (small red circles in Fig. 1). The rest of the rearranging particles did not rearrange again during our experiment.

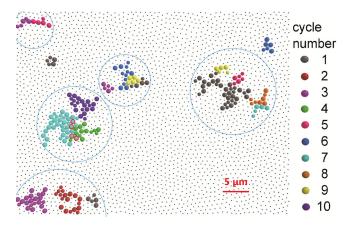


FIG. 1. Spatial distribution of rearranging clusters during thermal cycling. Rearranging clusters from different thermal cycle numbers shown by large spheres in different colors. Small dots are background particles that did not experience neighbor changes during the experiment. Large blue dotted circles indicate regions with repeated rearranging events. Small red circles show the only four overlapping rearranging particles during the experiment.

We examine the rearranging clusters in search of distinctive structural characteristics from the rest of the samples. We first measure the particle free volumes (V_f) . We employ radical Voronoi tessellation to define a polygon for each particle for our binary mixture [45-47]. The radical Voronoi tessellation avoids cutting through large spheres by taking into account the relative particle sizes. The V_f is determined by subtracting the cross-section area of a particle from the area of the polygon. The softsphere interaction between the colloidal particles allows negative V_f for compressed particles. The average V_f for particles in a rearranging cluster before and after local rearrangements is effectively the same as that of all the particles in the sample, with slightly larger widths, as plotted in Fig. 2(a). Figure 2(b) plots the measured rearranging probability as a function of initial V_f for individual particles. Particles with extremely high or extremely low V_f are those most likely to rearrange compared to particles with average V_f . The curve in Fig. 2(b) is roughly symmetric to the system average, suggesting that loose and compact regions are equally prone to rearrangement. This symmetry may explain why previous studies that searched for monotonic correlations between local free volumes and rearrangements did not find a significant overlap [14].

The rearranging probability for particles with the most extreme free volumes is about 3 times higher than that of the least active particles, which gives the free volume relatively weak ability to predict rearranging regions, compared to other known parameters such as soft spots

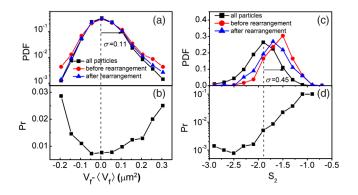


FIG. 2. Free volumes and local structural entropy in colloidal glasses. (a) Distribution of V_f for all particles (black squares), initial V_f for rearranging particles (red circles), and V_f of rearranging particles after deformations (blue triangles). (b) Rearranging probability as a function of initial V_f (subtracted by the average V_f). (c) S_2 distribution for all particles (black squares), for rearranging particles before deformations (blue triangles). (d) Rearranging probability as a function of initial S_2 . Dashed lines and arrows show the average value and standard deviation, respectively, of V_f and S_2 for the distributions for all particles. All distributions are obtained from the combined data of ten cycles and are individually normalized.

[12–14]. As a highly local structural parameter, particle free volume focuses only on the contacting neighbors of a particle. Rearrangements in glasses, however, require the cooperation between particles over a longer range [12,48]. Thus, the structural correlations beyond the first-neighbor shell need to be considered to better identify rearranging regions. Pair correlation function q(r) is a simple measure of structural correlations in glasses. Using local g(r), the structural entropy of a particle *i* can be defined as $S_{2,i} =$ $-1/2\sum_{\nu}\rho_{\nu}\int d\vec{r} \{g_{i}^{\mu\nu}(\vec{r})\ln g_{i}^{\mu\nu}(\vec{r}) - [g_{i}^{\mu\nu}(\vec{r}) - 1]\} \quad [15,28,49],$ where μ and ν denote the type of particles (large or small), ρ_{ν} is the number density of ν particles, and $g_{i}^{\mu\nu}(\vec{r})$ is the pair correlation function between particle *i* of type μ and particles of ν type. Structural entropy measures the loss of entropy due to positional correlations [50]; high S_2 values indicate less correlated local structures and vice versa. In our experiments, the integration is truncated at the third-neighbor shell to avoid the loss of a large fraction of particles due to the boundary effect. We verify, for particles near the center of the field of view, that the S_2 obtained from three shells of neighbors are highly correlated to S_2 obtained from five neighbor shell integrations, with correlation coefficients about 0.9. Beyond the fifth-neighbor shell, local q(r) becomes very close to 1, thus contributing only negligibly to measured S_2 . For each particle, time average is performed for local g(r) before integration to remove short-time fluctuations, although, in principle, this averaging is not critical to the measurements of S_2 [51]. We limit the averaging time to the last 60 s (1200 frames) of each thermal cycle, within the β relaxation time of the sample, to avoid a direct coupling to diffusive dynamics [29,51].

The structural entropy of rearranging particles are qualitatively different from those of the general population in the colloidal glass. Figure 2(c) plots the distribution of S_2 for all the particles in the colloidal glass (black squares) and those for rearranging particles (red circles). Compared to V_f distributions, the distribution of S_2 for rearranging particles peaks at a significantly higher value than the distribution from all particles, indicating less correlated structures of rearranging clusters. After rearrangements, the S_2 values of particles in rearranging clusters are decreased but remain elevated compared to the system average, as shown by blue triangles in Fig. 2(c). Figure 2(d) plots the rearranging probability as a function of initial S_2 for individual particles. The probability to rearrange increases almost monotonically with S_2 , with 2 orders of magnitude difference between the most stable particles and most particles, revealing intrinsically unstable different responses from local structures in colloidal glasses. Qualitatively the same correlations as shown in Fig. 2 have been observed in samples of different packing fractions and under different perturbation patterns [36].

Similar to soft modes, the structural entropy for individual particles can be employed to predict regions prone to rearrangements. The colored contour plot in Fig. 3(a) shows the initial distribution of S_2 in the colloidal glass *before* thermal cycling, and black circles overplotted on the contour plot are the rearranging clusters observed in the following ten cycles. Remarkable overlap is observed between regions with high S_2 values and rearranging clusters, which demonstrates that local S_2 is able to predict rearranging regions long before they actually occur in colloidal glasses under thermal cycling. This correlation is enhanced when the system is closer to the relaxation events, as shown in the video of the evolution of the S_2 distributions and the rearranging clusters from each thermal cycle in Supplemental Material [36].

The overlap between the S_2 distribution before thermal cycling and rearranging clusters in later cycles suggests that the distribution of high S_2 regions evolves slowly under thermal cycling, which can be clearly seen in the video in Supplemental Material [36]. An example of the stability

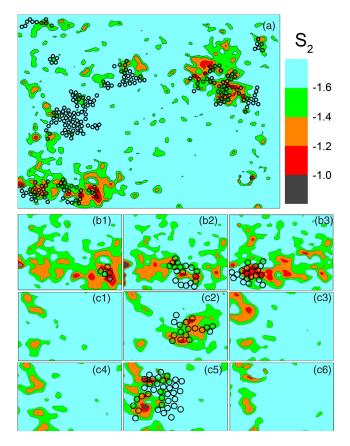


FIG. 3. Spatial distribution of particle S_2 and rearranging clusters. (a) Colored contour plot: S_2 distribution before thermal cycling and rearranging clusters observed in the following ten cycles (black circles). (b1)–(b3) Contour plots: local S_2 distributions before the first, second, and third thermal cycle in a subregion in the sample; black circles: rearranging clusters from first, second, and third in the same region. (c1)–(c6) Contour plots: local S_2 distributions from the third to eighth thermal cycle in another subregion in the sample; black circles: rearranging clusters from the fourth cycle (c2) and the seventh cycle (c5).

of local distribution of S_2 is shown in Figs. 3(b1)–3(b3) (bottom-left corner of the video in Supplemental Material [36]). During the three consecutive local rearrangements, the area with high S_2 values only changes slightly. On the other hand, high S_2 regions can be created through local fluctuations. Some rearranging clusters that have small overlaps with high S_2 regions in Fig. 3(a) are found to significantly overlap with high S_2 spots that emerge during the thermal cycling. An example is shown in Figs. 3(c1)-3(c6) (mid-left region of the video in Supplemental Material [36]). Before the fourth and seventh cycles, two high S_2 spots are developed with no preceding rearranging clusters in its immediate neighborhood. The emergences of these two spots are followed by two rearranging events in the same region, after which local S_2 distribution falls back to the background.

We quantitatively evaluate the correlations between high S_2 regions and local rearrangements during thermal cycling. High S_2 clusters are identified to the particle level by applying a cutoff to the distribution of local structural entropy. Averaged correlation between high S_2 clusters and rearranging clusters is measured to be 0.34 for a cutoff value of -1.5 [36], comparable to the correlations between soft spots and local rearrangements in colloidal experiments [13]. The highest single-cycle correlation is measured to be 0.69 [36]. We also measure the direct correlations between nonaffine displacement coefficient D_{\min}^2 and S_2 for all particles, regardless of neighbor changes. D_{\min}^2 measures the particle level nonaffine strain and is defined as the minimum of $D^2(t_1, t_2) =$ $\sum_{n} \sum_{i} [r_{n,t_2}^i - r_{0,t_2}^i - \sum_{j} (\delta_{ij} + \varepsilon_{ij}) \times (r_{n,t_1}^j - r_{0,t_1}^j)]^2, \text{ where } r_{n,t}^i \text{ is the } i\text{th } (x \text{ or } y) \text{ component of the position of the }$ *n*th particle at time t (before or after a single cycle) and index n runs over the particles within the interaction range of the reference particle n = 0. The ε_{ij} that minimizes D^2 is calculated based on $r_{n,t}^i$ [10]. Direct correlation between D_{\min}^2 and particle S_2 for all particles at each cycle yields an averaged correlation coefficient of 0.30 [36], suggesting that particles with higher S_2 values tend to experience larger strains under external loading. The noise levels for the above two correlation coefficients are both on the order of $\sim 1/\sqrt{N}$, where N is the number of particles in the field of view.

On the other hand, structurally stable regions associated with low S_2 values in the colloidal glasses show a strong resistance to the migration of rearranging clusters. Figure 4 plots the particles whose S_2 remain below -1.6 (the 90th percentile in the S_2) for the duration of the experiment (large gray circles). These gray particles form a large percolating network that only overlaps with rearranging clusters on the boundaries with high S_2 areas (small blue dots). The structural stability of these correlated domains may be related to the amorphously ordered structures predicted by random first-order transitions theories [24,25] and the dynamical slowdown during glass transitions

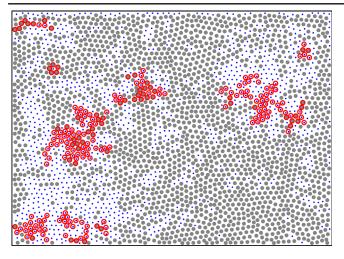


FIG. 4. Stable regions in colloidal glasses. Spatial distributions of particles with S_2 values below -1.6 during the experiment (large filled gray circles), particles with S_2 values exceeding -1.6 for at least one cycle (small blue dots), and rearranging clusters (red empty circles).

[28,29,52], which are fundamental challenges in the physics of glasses. And our results suggest S_2 may be employed to search for such structures.

We can now make the analogy between structural defects in crystals to high S_2 regions in colloidal glasses. In polycrystalline solids, structural correlation is strong between atoms belonging to the same domain and weak for atoms on defect sites. Similarly, the high S_2 spots in colloidal glasses are structurally less correlated compare to the percolating low S_2 domains. Thus, high S_2 regions can be viewed as equivalent "structural defects" in glasses. This analogy is supported by the mechanical susceptibility of these regions and their creation and migration dynamics observed in the experiments.

In summary, we measure the structures of local rearranging clusters in colloidal glasses under thermal cycling. The local structural entropy of the rearranging regions is distinctively higher than that of the more stable background in glasses. Equivalent defects, that share the essential characteristics of structural defects in crystals, can be defined by the high S_2 values. As a structural parameter, S_2 provides a sensible measure of the apparently disordered structures of glasses. The strong connections between the structural entropy to physical observables such as system dynamics [28,49,53] and local relaxations suggest that structural correlations may play a critical role in the physics of glasses. An investigation of more complex structural correlations such as multibody correlations [51] in glasses may reveal deeper connections between structures and the macroscopic properties of glasses.

We thank Yiwu Zong, Baoan Sun, and Chris H. Rycroft for instructive discussions. This work was supported by the MOST 973 Program (No. 2015CB856800). K. C. also acknowledges the support from the NSFC (No. 11474327). ^{*}whw@iphy.ac.cn [†]kechen@iphy.ac.cn

- H. W. Sheng, W. K. Luo, F. M. Alamgir, J. M. Bai, and E. Ma, Nature (London) 439, 419 (2006).
- [2] T. Kawasaki, T. Araki, and H. Tanaka, Phys. Rev. Lett. 99, 215701 (2007).
- [3] A. Hirata, L. J. Kang, T. Fujita, B. Klumov, K. Matsue, M. Kotani, A. R. Yavari, and M. W. Chen, Science 341, 376 (2013).
- [4] Z. W. Wu, M. Z. Li, W. H. Wang, and K. X. Liu, Nat. Commun. 6, 6035 (2015).
- [5] Y. C. Hu, F. X. Li, M. Z. Li, H. Y. Bai, and W. H. Wang, Nat. Commun. 6, 8310 (2015).
- [6] X. Yang, C. Zhou, Q. Sun, L. Hu, J. C. Mauro, C. Wang, and Y. Yue, J. Phys. Chem. B 118, 10258 (2014).
- [7] A. V. Anikeenko and N. N. Medvedev, Phys. Rev. Lett. 98, 235504 (2007).
- [8] Z. Lu, W. Jiao, W. H. Wang, and H. Y. Bai, Phys. Rev. Lett. 113, 045501 (2014).
- [9] Z. Wang, B. A. Sun, H. Y. Bai, and W. H. Wang, Nat. Commun. 5, 5823 (2014).
- [10] M. L. Falk and J. S. Langer, Phys. Rev. E 57, 7192 (1998).
- [11] A. Argon, Acta Metall. 27, 47 (1979).
- [12] A. Widmer-Cooper, H. Perry, P. Harrowell, and D. R. Reichman, Nat. Phys. 4, 711 (2008).
- [13] K. Chen, M. L. Manning, P. J. Yunker, W. G. Ellenbroek, Z. Zhang, A. J. Liu, and A. G. Yodh, Phys. Rev. Lett. 107, 108301 (2011).
- [14] M. L. Manning and A. J. Liu, Phys. Rev. Lett. 107, 108302 (2011).
- [15] H. Tong and N. Xu, Phys. Rev. E 90, 010401 (2014).
- [16] J. Ding, S. Patinet, M. L. Falk, Y. Cheng, and E. Ma, Proc. Natl. Acad. Sci. U.S.A. 111, 14052 (2014).
- [17] P. Sollich, F. Lequeux, P. Hébraud, and M. E. Cates, Phys. Rev. Lett. 78, 2020 (1997).
- [18] F. Spaepen, Acta Metall. 25, 407 (1977).
- [19] T. Egami, K. Maeda, and V. Vitek, Philos. Mag. A 41, 883 (1980).
- [20] T. Egami, Prog. Mater. Sci. 56, 637 (2011).
- [21] P. Y. Huang, S. Kurasch, J. S. Alden, A. Shekhawat, A. A. Alemi, P. L. McEuen, J. P. Sethna, U. Kaiser, and D. A. Muller, Science 342, 224 (2013).
- [22] E. D. Cubuk, S. S. Schoenholz, J. M. Rieser, B. D. Malone, J. Rottler, D. J. Durian, E. Kaxiras, and A. J. Liu, Phys. Rev. Lett. 114, 108001 (2015).
- [23] S. S. Schoenholz, E. D. Cubuk, D. M. Sussman, E. Kaxiras, and A. J. Liu, Nat. Phys. **12**, 469 (2016).
- [24] V. Lubchenko and P. G. Wolynes, Annu. Rev. Phys. Chem. 58, 235 (2007).
- [25] G. Parisi and F. Zamponi, Rev. Mod. Phys. 82, 789 (2010).
- [26] G. Biroli, J.-P. Bouchaud, A. Cavagna, T. S. Grigera, and P. Verrocchio, Nat. Phys. 4, 771 (2008).
- [27] A. S. Keys, A. R. Abate, S. C. Glotzer, and D. J. Durian, Nat. Phys. 3, 260 (2007).
- [28] H. Tanaka, T. Kawasaki, H. Shintani, and K. Watanabe, Nat. Mater. 9, 324 (2010).
- [29] Z. Zheng et al., Nat. Commun. 5, 3829 (2014).
- [30] L. Berthier and R. L. Jack, Phys. Rev. E 76, 041509 (2007).
- [31] P. Schall, D. A. Weitz, and F. Spaepen, Science 318, 1895 (2007).

- [32] R. Candelier, O. Dauchot, and G. Biroli, Phys. Rev. Lett. 102, 088001 (2009).
- [33] N.C. Keim and P.E. Arratia, Soft Matter 9, 6222 (2013).
- [34] I. Regev, T. Lookman, and C. Reichhardt, Phys. Rev. E 88, 062401 (2013).
- [35] D. Fiocco, G. Foffi, and S. Sastry, Phys. Rev. E 88, 020301 (2013).
- [36] See Supplemental Material at http://link.aps.org/ supplemental/10.1103/PhysRevLett.116.238003 for a discussion of additional experimental details.
- [37] P. J. Yunker, K. Chen, M. D. Gratale, M. A. Lohr, T. Still, and A. G. Yodh, Rep. Prog. Phys. 77, 056601 (2014).
- [38] T. Still, K. Chen, A. M. Alsayed, K. B. Aptowicz, and A. G. Yodh, J. Colloid Interface Sci. 405, 96 (2013).
- [39] K. Chen et al., Phys. Rev. Lett. 105, 025501 (2010).
- [40] K. Chen, T. Still, S. Schoenholz, K. B. Aptowicz, M. Schindler, A. C. Maggs, A. J. Liu, and A. G. Yodh, Phys. Rev. E 88, 022315 (2013).
- [41] Y. Han, N. Y. Ha, A. M. Alsayed, and A. G. Yodh, Phys. Rev. E 77, 041406 (2008).

- [42] Z. Wang, F. Wang, Y. Peng, Z. Zheng, and Y. Han, Science 338, 87 (2012).
- [43] P. J. Yunker, Z. Zhang, K. B. Aptowicz, and A. G. Yodh, Phys. Rev. Lett. 103, 115701 (2009).
- [44] J. C. Crocker and D. G. Grier, J. Colloid Interface Sci. 179, 298 (1996).
- [45] S. C. Zhao, S. Sidle, H. L. Swinney, and M. Schröter, Europhys. Lett. 97, 34004 (2012).
- [46] S. Sastry, D. S. Corti, P. G. Debenedetti, and F. H. Stillinger, Phys. Rev. E 56, 5524 (1997).
- [47] C. H. Rycroft, Chaos 19, 041111 (2009).
- [48] A. Widmer-Cooper and P. Harrowell, J. Non-Cryst. Solids 352, 5098 (2006).
- [49] A. Samanta, Sk. Musharaf Ali, and S. K. Ghosh, Phys. Rev. Lett. 92, 145901 (2004).
- [50] D.C. Wallace, J. Chem. Phys. 87, 2282 (1987).
- [51] M. Leocmach, J. Russo, and H. Tanaka, J. Chem. Phys. 138, 12A536 (2013).
- [52] Y. Zhou and S. T. Milner, Soft Matter 11, 2700 (2015).
- [53] J. Mittal, J. R. Errington, and T. M. Truskett, J. Phys. Chem. B 110, 18147 (2006).