## Superconductivity Induced by Site-Selective Arsenic Doping in Mo<sub>5</sub>Si<sub>3</sub>

Bin-Bin Ruan,\* Jun-Nan Sun, Meng-Hu Zhou, Qing-Song Yang, Ya-Dong Gu, Gen-Fu Chen, Lei Shan, and Zhi-An Ren\*



ABSTRACT: Arsenic doping in silicides has been much less studied compared with phosphorus. In this study, superconductivity is successfully induced by As doping in  $Mo_sSi_3$ . The superconducting transition temperature ( $T_c$ ) reaches 7.7 K, which is higher than those in previously known W<sub>5</sub>Si<sub>3</sub>-type superconductors. Mo<sub>5</sub>Si<sub>2</sub>As is a type-II BCS superconductor with upper and lower critical fields of 6.65 T and 22.4 mT, respectively. In addition, As atoms are found to selectively take the 8h sites in Mo<sub>5</sub>Si<sub>2</sub>As. The emergence of superconductivity is possibly due to the shift of Fermi level as a consequence of As doping, as revealed by the specific heat measurements and first-principles calculations. Our work provides not only another example of As doping but also a practical strategy to achieve superconductivity in silicides through Fermi level engineering.

rsenic doping in elemental silicon has been known for Amore than half a century.<sup>1</sup> And the kinetics for As diffusion in silicon had been well established.<sup>2,3</sup> However, reports on arsenic doping in silicides are unexpectedly scarce. To the best of our knowledge, there are only 5 examples in bulk materials, namely:  $Cr_3Si_{1-x}As_x^4$  Fe<sub>5</sub>SiAs,<sup>5</sup> ZrSi<sub>1-x</sub>As<sub>x</sub>Te,<sup>6</sup> and  $Zr(Hf)As_{2-x}Si_x$ .

On the other hand, phosphorus doping had been reported in more than 30 silicides:  $CoSi_{0.4}P_{0.6,7}$   $USi_{0.17}P_{0.83,7}$   $^{10}$  Fe<sub>5</sub>SiP,<sup>5</sup> Nb<sub>5</sub>Si<sub>3-x</sub>P<sub>0.5+x</sub><sup>11</sup> Gd<sub>5</sub>Si<sub>4-x</sub>P<sub>x</sub><sup>12</sup> and MSi<sub>x</sub>P<sub>y</sub> (M = Fe, Co, Ru, Rh, Pd, Os, Ir, and Pt, and  $x + y \ge 4$ ), <sup>13,14</sup> to name a few. Compared to the P dopant, the much less studied As doping offers an opportunity to chase for new compounds in this field.

Our group has been working on MoAs-based superconductors, and has discovered a series of quasi-one-dimensional superconductors  $A_2Mo_3As_3$  (A = K, Rb, Cs).<sup>15–17</sup> Recently, superconductivity was reported in Re-doped Mo<sub>5</sub>Si<sub>3</sub> with a transition temperature  $(T_c)$  of 5.8 K, setting a new record in the isostructural compounds.<sup>18</sup> We, thus, naturally conducted a systematic As doping study on Mo<sub>5</sub>Si<sub>3</sub>, which led to the discovery of superconductivity in Mo<sub>5</sub>Si<sub>3-x</sub>As<sub>x</sub>.

Mo<sub>5</sub>Si<sub>3</sub> crystallizes in a tetragonal W<sub>5</sub>Si<sub>3</sub>-type structure (space group *I4/mcm*), as illustrated in Figure 1a. Notice that there are two kinds of Si atoms taking different Wyckoff positions: Si1 at the 4a sites, and Si2 at the 8h sites. In this study, we show that arsenic can be doped into Mo<sub>5</sub>Si<sub>3</sub>. Interestingly, As atoms selectively takes the Si2 (8h) sites. Furthermore, superconductivity was observed in Mo<sub>5</sub>Si<sub>3-x</sub>As<sub>x</sub>  $(x \ge 0.5)$ , with  $T_c$  increases from 4.4 K in Mo<sub>5</sub>Si<sub>2.5</sub>As<sub>0.5</sub> to 7.7 K in Mo<sub>5</sub>Si<sub>2</sub>As. The data for Mo<sub>5</sub>Si<sub>2</sub>As are shown in the main text, while additional information about  $Mo_5Si_{3-x}As_x$  ( $x \neq 1.0$ ) is shown in Figures S2–S4.

Figure 1c demonstrates the powder X-ray diffraction (XRD) pattern of Mo<sub>5</sub>Si<sub>2</sub>As. For Mo<sub>5</sub>Si<sub>2</sub>As, there are at least two configurations (configs.) in which As atoms take different sites (Figure 1b). In config. 1, As atoms randomly take the Si2 (8h) sites. While in config. 2, As atoms take all the Si1 (4a) sites. However, the observed XRD pattern can only be refined with config. 1. In particular, any occupation of As at the Si1 (4a)sites will cause a significant enhancement of the (200) peak, which prevents the refinement from convergence. (See Figure S5.) A brief list of the refined crystallographic parameters is shown in Table 1. More details about the refinement results can be found in Table S1. According to the refinement, there is about 10.3 wt % of Mo<sub>3</sub>Si impurity in the sample. The refined composition is Mo<sub>5.00(1)</sub>Si<sub>1.97(3)</sub>As<sub>1.05(8)</sub>, which is close to that determined by energy-dispersive X-ray spectroscopy. (For the sample morphology and elemental mapping, see Figure S6.)

The site-selective doping of As is further backed by the firstprinciples calculations. As shown in Figure 1b, the enthalpy (H) for config. 1 is lower compared with config. 2, which means As doping at the Si2 (8h) sites is more favorable in energy. The above conclusion is based on zero temperature calculations, but it is also true for finite temperatures. This is because the free energy G = H - TS, and the entropy S for config. 1 is larger. (Notice the Boltzmann relation  $S = k_B \ln \Omega$ , where  $k_{\rm B}$  is the Boltzmann constant, and  $\Omega$  is the number of microstates)

Figure 2a shows the temperature dependence of resistivity  $(\rho)$  of Mo<sub>5</sub>Si<sub>2</sub>As. The monotonous decrease of  $\rho$  upon cooling indicates a metallic nature. Superconducting transition is observed below 7.7(1) K ( $T_c^{onset}$ ), with zero resistivity achieved at 7.4(1) K ( $T_c^{zero}$ ). Figure 2b emphasizes the region of the

Received: May 12, 2022 Published: June 28, 2022







Figure 1. (a) Crystal structure of  $Mo_5Si_3$ . (b) Two possible configurations of  $Mo_5Si_2As$  and the corresponding enthalpies from DFT calculations. (c) Room-temperature XRD pattern of  $Mo_5Si_2As$  and its Rietveld refinement. The vertical bars indicate the Bragg positions for  $Mo_5Si_2As$  and the  $Mo_3Si$  impurity. The relevant (200) peak of  $Mo_5Si_2As$  is marked.

superconducting transition. Upon the application of the magnetic field, the transition is gradually suppressed. The temperature dependence of upper critical field  $(\mu_0 H_{c2}(T))$  can thus be determined, which is shown in the inset of Figure 2a. A Ginzburg–Landau (G–L) fit gives  $\mu_0 H_{c2}(0) = 6.65(4)$  T.

DC magnetic susceptibility  $(4\pi\chi)$  of Mo<sub>5</sub>Si<sub>2</sub>As from 1.8 to 10.0 K under 10 Oe is demonstrated in Figure 2c. Bulk superconductivity is confirmed in  $4\pi\chi(T)$  curves, with large diamagnetic signals observed below 7.4 K. The value of  $T_c$ determined from  $4\pi\chi(T)$  agrees very well with that from  $\rho(T)$ . We also measured the isothermal magnetization curves for Mo<sub>5</sub>Si<sub>2</sub>As. The results are shown in Figure 2(d). The lower critical fields ( $\mu_0H_{c1}(T)$ ) are determined from the deviation of the curves from initial Meissner states. As shown in the inset of Figure 2(c),  $\mu_0H_{c1}(T)$  can be fitted with the G–L relation:  $\mu_0H_{c1}(T) = \mu_0H_{c1}(0)[1 - (T/T_c)^2]$ , giving  $\mu_0H_{c1}(0) =$ 22.4(5) mT.

A series of superconducting parameters can be determined using  $\mu_0 H_{c2}(0)$  and  $\mu_0 H_{c1}(0)$ . The values of these parameters are listed in Table 2. The definitions and calculation details for them can be found in Supporting Information. Notice



**Figure 2.** (a) Temperature dependence of resistivity ( $\rho$ ) of Mo<sub>5</sub>Si<sub>2</sub>As under zero magnetic field. Inset shows the temperature dependence of upper critical field. (b) Zoom-in of the superconducting transition on  $\rho(T)$  under magnetic fields from 0 to 6.5 T. (c) DC magnetic susceptibility of Mo<sub>5</sub>Si<sub>2</sub>As under 10 Oe. Inset shows the temperature dependence of lower critical field. (d) Isothermal magnetization at different temperatures.

 $\kappa_{GL} \gg 1/\sqrt{2}$ , suggesting type-II superconductivity in Mo<sub>5</sub>Si<sub>2</sub>As.

Table 2. Superconducting and Thermodynamic Parameters of  $Mo_{s}Si_{2}As$ 

parameter (unit)	value
$T_{\rm c}^{\rm onset}$ (K)	7.7(1)
$T_{\rm c}^{\rm zero}$ (K)	7.4(1)
$\mu_0 H_{c1}(0) \ (mT)$	22.4(5)
$\mu_0 H_{c2}(0)$ (T)	6.65(4)
$\mu_0 H_c(0)$ (T)	0.22(0)
$\xi_{GL}$ (nm)	7.03(2)
$\lambda_{GL}$ (nm)	159.5(4)
$\kappa_{GL}$	22.7(1)
$\gamma \text{ (mJ mol}^{-1} \text{ K}^{-2})$	34.74(8)
$\beta$ (mJ mol <sup>-1</sup> K <sup>-4</sup> )	0.273(1)
$\Theta_D$ (K)	385(2)
$\lambda_{ep}$	0.66(1)
$\Delta C_e / \gamma T_c$	1.46(2)

The specific heat  $(C_p)$  of  $Mo_5Si_2As$  was measured to investigate the superconducting, as well as the thermodynamic

Table 1. Crystallographic Parameters of  $Mo_5Si_2As$  from Rietveld Refinement of XRD ( $R_p = 1.03\%$ ,  $R_{wp} = 1.49\%$ )

atom (site)	x	у	z	$U_{eq}$ (0.01 Å <sup>2</sup> )	occupancy
Mo1 (4 <i>b</i> )	0	0.5	0.25	0.137(27)	0.987(2)
Mo2 (16k)	0.07652(4)	0.22196(5)	0	0.392(17)	1.0
Si1 (4 <i>a</i> )	0	0	0.25	0.84(15)	1.0
Si2/As (8h)	0.16592(8)	0.66592(8)	0	1.36(6)	0.474(4)/0.526

properties. The results are shown in Figure 3(a). Notice the data have been corrected to exclude the influence of Mo<sub>3</sub>Si



**Figure 3.** (a) Temperature dependence of specific heat  $(C_p)$  of  $Mo_5Si_2As$  under zero magnetic field, and under a field of 8 T. The  $C_p$  data for  $Mo_5Si_3$  are also plotted for comparison. Solid lines are fittings with Debye model. (b) Temperature dependence of the electronic contribution of  $C_p$  under zero magnetic field. Parts a and b share the same *y*-axis range. (c) Calculated DOS for  $Mo_5Si_3$  and  $Mo_5Si_2As$  near the Fermi level.

impurity. Thermodynamic parameters for Mo<sub>3</sub>Si were adopted from the literature.<sup>19</sup> Under zero magnetic field,  $C_p$  for  $Mo_5Si_2As$  clearly shows an anomaly at ~7.0 K, validating the bulk superconductivity. When a field of 8 T was applied, superconductivity was completely suppressed. The data under 8 T can be well fitted with the Debye model:  $C_p(T) = \gamma T +$  $\beta T^3$ , giving  $\gamma = 34.74(8)$  mJ mol<sup>-1</sup> K<sup>-2</sup>, and  $\beta = 0.273(1)$  mJ mol<sup>-1</sup> K<sup>-4</sup>. The Debye temperature ( $\Theta_D$ ) is thus determined by  $\Theta_D = (12\pi^4 NR/5\beta)^{1/3}$  to be 385(2) K. (N is the number of atoms in a formula unit (f.u.), and R is the ideal gas constant.) The electronic contribution of specific heat  $(C_e)$  is extracted by  $C_e = C_p|_{\mu_0 H=0} - C_p|_{\mu_0 H=8T} + \gamma$ , which is shown in Figure 3b.  $C_e$  at the superconducting state can be well fitted with the so-called  $\alpha$ -model.<sup>20</sup> The normalized  $C_e$  change  $\Delta C_e/$  $\gamma T_c = 1.46(2)$ , which is close to the BCS weak-coupling ratio (1.43). The electron-phonon coupling strength  $(\lambda_{ep})$  can be estimated by using the inverted McMillan formula:

$$\lambda_{ep} = \frac{1.04 + \mu^* \ln(\Theta_D / 1.45T_c)}{(1 - 0.62\mu^*) \ln(\Theta_D / 1.45T_c) - 1.04}$$
(1)

By setting the Coulomb screening parameter  $\mu^* = 0.13$ , a typical value for intermetallics, we get  $\lambda_{ep} = 0.66(1)$ . These results suggest a weak to moderate coupling in Mo<sub>5</sub>Si<sub>2</sub>As.

For comparison, we also measured the temperature dependence of  $C_p$  of Mo<sub>5</sub>Si<sub>3</sub>, which is shown in Figure 3a.  $\gamma$  and  $\Theta_D$  for Mo<sub>5</sub>Si<sub>3</sub> are estimated to be 19.80 mJ mol<sup>-1</sup> K<sup>-2</sup> and 659 K, respectively. These values are comparable with previous report on Mo<sub>5</sub>Si<sub>3</sub> single crystals.<sup>22</sup> Notice that  $\gamma$  in Mo<sub>5</sub>Si<sub>2</sub>As is

Communication adoped Mo<sub>5</sub>Si<sub>3</sub>, and the larg

significantly larger than the undoped Mo<sub>S</sub>Si<sub>3</sub>, and the large decrease of  $\Theta_D$  suggests substantial softening of the lattice. To take more insight into this, the density of states (DOS) was calculated for Mo<sub>S</sub>Si<sub>3</sub> and Mo<sub>S</sub>Si<sub>2</sub>As. The results are shown in Figure 3c. One may notice the similar shapes of the DOS curves, which means that the bands can be considered rigid in our case. (For the band structures, see Figure S7) Arsenic doping shifts the Fermi level ( $E_F$ ) to higher energy, causing a significant enhancement of DOS at  $E_F$  ( $N(E_F)$ ). This is as expected, since As hosts one more valence electron compared with Si.  $N(E_F)$  for Mo<sub>S</sub>Si<sub>3</sub> is 4.13 eV<sup>-1</sup> f.u.<sup>-1</sup>, while  $N(E_F)$  for Mo<sub>S</sub>Si<sub>2</sub>As is 7.37 eV<sup>-1</sup> f.u.<sup>-1</sup>. The enhancement of  $N(E_F)$  led to the increase of  $\gamma$ . We can theoretically estimate the value of  $\gamma$ for Mo<sub>S</sub>Si<sub>2</sub>As by

pubs.acs.org/IC

$$\gamma = \frac{1}{3} N(E_F) \pi^2 k_B^2 (1 + \lambda_{ep})$$
<sup>(2)</sup>

to be ~29.02 mJ mol<sup>-1</sup> K<sup>-2</sup>, which agrees well with the experimental value. According to McMillan's formalism,  $\lambda_{ep} = [N(E_F)\langle I^2\rangle]/[M\langle\omega^2\rangle]$ , where *M* is the atomic mass,  $\langle I^2\rangle$  and  $\langle\omega^2\rangle$  stand for averages of the squared electronic matrix elements on the Fermi surface, and of the squared phonon frequencies, respectively.<sup>21</sup> The emergence of superconductivity in Mo<sub>5</sub>Si<sub>3-x</sub>As<sub>x</sub> should be due to the enhancement of  $N(E_F)$ , and possibly the softening of lattice (as evidenced by the large decrease of  $\Theta_D$ ).

Lastly, we would like to point out that a  $T_c$  of 7.7 K is fairly high in W<sub>5</sub>Si<sub>3</sub>-type superconductors. A comparison of  $T_c$ between Mo<sub>5</sub>Si<sub>2</sub>As and previously reported W<sub>5</sub>Si<sub>3</sub>-type superconductors is illustrated in Figure 4.<sup>18,23–30</sup> Notice that



**Figure 4.** Evolution of  $T_c$  in W<sub>5</sub>Si<sub>3</sub>-type superconductors upon the valence electron per atom (e/a). Note that the peak at  $e/a \sim 5.4$  is just a tentative guide.

maximal  $T_c$  occurs when the average number of valence electrons per atom  $(e/a) \sim 4.6$  or 5.4. Here, the second peak (at  $e/a \sim 5.4$ ) is just a tentative guide, and is less reliable as there are only a few examples with e/a > 5. The phenomenon that  $T_c$  peaks at certain e/a values, known as the Matthias rule, has also been observed in other intermetallics.<sup>31</sup> Currently, the region of e/a > 5.5 remains unexplored, so more superconductors may be discovered if more electrons can be introduced into this structural family.

In summary, we report the discovery of superconductivity in  $Mo_5Si_{3-x}As_x$  (0.5  $\leq x \leq 1.25$ ), in which a maximal  $T_c$  of 7.7 K

is observed in  $Mo_5Si_2As$ . Arsenic doping in  $Mo_5Si_3$  is found to be site-selective. According to specific heat measurements and first-principles calculations, the emergence of superconductivity is related to the shift of Fermi level. Our method of As doping should be generally applicable to other silicides, in which more superconductors can be expected.

#### ASSOCIATED CONTENT

#### **Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acs.inorgchem.2c01647.

Experimental and calculation details, evolution of lattice parameters and superconducting properties of  $Mo_5Si_{3-x}As_x$  ( $0 \le x \le 1.25$ ), discussion on the siteselective doping in  $Mo_5Si_2As$ , detailed XRD refinement results, SEM images, elemental mapping, and band structure for  $Mo_5Si_2As$  (PDF)

#### AUTHOR INFORMATION

#### Corresponding Authors

- Bin-Bin Ruan Institute of Physics and Beijing National Laboratory for Condensed Matter Physics, Chinese Academy of Sciences, Beijing 100190, China; orcid.org/0000-0003-4642-7782; Email: bbruan@mail.ustc.edu.cn
- Zhi-An Ren Institute of Physics and Beijing National Laboratory for Condensed Matter Physics, Chinese Academy of Sciences, Beijing 100190, China; School of Physical Sciences, University of Chinese Academy of Sciences, Beijing 100049, China; orcid.org/0000-0003-4308-7372; Email: renzhian@iphy.ac.cn

#### Authors

Jun-Nan Sun – Information Materials and Intelligent Sensing Laboratory of Anhui Province, Institutes of Physical Science and Information Technology, Anhui University, Hefei 230601, China; Key Laboratory of Structure and Functional Regulation of Hybrid Materials (Anhui University), Ministry of Education, Hefei 230601, China

Meng-Hu Zhou – Institute of Physics and Beijing National Laboratory for Condensed Matter Physics, Chinese Academy of Sciences, Beijing 100190, China

**Qing-Song Yang** – Institute of Physics and Beijing National Laboratory for Condensed Matter Physics, Chinese Academy of Sciences, Beijing 100190, China; School of Physical Sciences, University of Chinese Academy of Sciences, Beijing 100049, China

Ya-Dong Gu – Institute of Physics and Beijing National Laboratory for Condensed Matter Physics, Chinese Academy of Sciences, Beijing 100190, China; School of Physical Sciences, University of Chinese Academy of Sciences, Beijing 100049, China

Gen-Fu Chen – Institute of Physics and Beijing National Laboratory for Condensed Matter Physics, Chinese Academy of Sciences, Beijing 100190, China; School of Physical Sciences, University of Chinese Academy of Sciences, Beijing 100049, China; Songshan Lake Materials Laboratory, Dongguan, Guangdong 523808, China

Lei Shan – Information Materials and Intelligent Sensing Laboratory of Anhui Province, Institutes of Physical Science and Information Technology, Anhui University, Hefei 230601, China; Key Laboratory of Structure and Functional Regulation of Hybrid Materials (Anhui University), Ministry of Education, Hefei 230601, China

Complete contact information is available at: https://pubs.acs.org/10.1021/acs.inorgchem.2c01647

#### Notes

The authors declare no competing financial interest.

#### ACKNOWLEDGMENTS

This work was supported by the National Key Research and Development of China (Grant Nos. 2018YFA0704200, 2021YFA1401800, 2018YFA0305602, and 2017YFA0302904), the National Natural Science Foundation of China (Grant Nos. 12074414, 12074002, and 11774402), and the Strategic Priority Research Program of Chinese Academy of Sciences (Grant No. XDB25000000).

#### REFERENCES

(1) Trumbore, F. A. Solid solubilities of impurity elements in germanium and silicon. *Bell System Technical Journal* **1960**, 39, 205–233.

(2) Masetti, G.; Severi, M.; Solmi, S. Modeling of carrier mobility against carrier concentration in arsenic-, phosphorus-, and boron-doped silicon. *IEEE Trans. Electron Devices* **1983**, *30*, 764–769.

(3) Ramamoorthy, M.; Pantelides, S. T. Complex dynamical phenomena in heavily arsenic doped silicon. *Phys. Rev. Lett.* **1996**, 76, 4753.

(4) Boller, H.; Wolfsgruber, H.; Nowotny, H. Röntgenographische Untersuchungen in den Systemen Chrom-Silicium (Germanium)-Arsen. *Monatsh. Chem.* **1967**, *98*, 2356–2361.

(5) Ellner, M.; El-Boragy, M. Über die eisenhaltigen vertreter des strukturtyps Pd<sub>3</sub>Sb<sub>2</sub>. *J. Alloys Compd.* **1992**, *184*, 131–138.

(6) Wang, C.; Hughbanks, T. Main group element size and substitution effects on the structural dimensionality of zirconium tellurides of the ZrSiS type. *Inorg. Chem.* **1995**, *34*, 5524–5529.

(7) Grosvenor, A. P.; Cavell, R. G.; Mar, A.; Blyth, R. I. Analysis of the electronic structure of  $Hf(Si_{0.5}As_{0.5})As$  by X-ray photoelectron and photoemission spectroscopy. *J. Solid State Chem.* **2007**, *180*, 2670–2681.

(8) Gaultois, M. W.; Grosvenor, A. P.; Blanchard, P. E.; Mar, A. Ternary arsenides  $Zr(Si_xAs_{1-x})As$  with PbCl<sub>2</sub>-type ( $0 \le x \le 0.4$ ) and PbFCl-type (x = 0.6) structures. *J. Alloys Compd.* **2010**, 492, 19–25.

(9) Rundqvist, S.; Sillén, L.; Timm, D.; Motzfeldt, K.; Theander, O.; Flood, H. Phosphides of the B31 (MnP) structure type. *Acta Chem. Scand.* **1962**, *16*, 287–292.

(10) Adachi, H.; Imoto, S. Energy Band Structure of Uranium Compounds with NaCl Type Structure. *J. Nucl. Sci. Technol.* **1969**, *6*, 371–379.

(11) Lomnytska, Y. F.; Oryshchyn, S.; Kuz'ma, Y. B.; Guerin, R. Crystal Structure of  $Nb_3Si_{3-x}P_{0.5+x}$  (x = 0–0.5). *Inorg. Mater.* 2005, 41, 1067–1072.

(12) Svitlyk, V.; Miller, G. J.; Mozharivskyj, Y.  $Gd_SSi_{4-x}P_x$ : Targeted Structural Changes through Increase in Valence Electron Count. *J. Am. Chem. Soc.* **2009**, 131, 2367–2374.

(13) Kirschen, M.; Vincent, H.; Perrier, C.; Chaudouet, P.; Chenevier, B.; Madar, R. Synthesis and crystal structure of rhodium and iridium new phospho-silicides. *Mater. Res. Bull.* **1995**, *30*, 507–513.

(14) Perrier, C.; Kreisel, J.; Vincent, H.; Chaix-Pluchery, O.; Madar, R. Synthesis, crystal structure, physical properties and Raman spectroscopy of transition metal phospho-silicides  $MSi_xP_y$  (M = Fe, Co, Ru, Rh, Pd, Os, Ir, Pt). *J. Alloys Compd.* **1997**, 262–263, 71–77. (15) Mu, Q.-G.; Ruan, B.-B.; Zhao, K.; Pan, B.-J.; Liu, T.; Shan, L.; Chen, G.-F.; Ren, Z.-A. Superconductivity at 10.4 K in a novel quasione-dimensional ternary molybdenum pnictide K<sub>2</sub>Mo<sub>3</sub>As<sub>3</sub>. *Sci. Bull.* **2018**, 63, 952–956.

(16) Zhao, K.; Mu, Q.-G.; Ruan, B.-B.; Zhou, M.-H.; Yang, Q.-S.; Liu, T.; Pan, B.-J.; Zhang, S.; Chen, G.-F.; Ren, Z.-A. A New Quasi-One-Dimensional Ternary Molybdenum Pnictide  $Rb_2Mo_3As_3$  with Superconducting Transition at 10.5 K. *Chin. Phys. Lett.* **2020**, *37*, 097401.

(17) Zhao, K.; Mu, Q.-G.; Ruan, B.-B.; Liu, T.; Pan, B.-J.; Zhou, M.-H.; Zhang, S.; Chen, G.-F.; Ren, Z.-A. Synthesis and superconductivity of a novel quasi-one-dimensional ternary molybdenum pnictide  $Cs_2Mo_3As_3$ . *APL Mater.* **2020**, *8*, 031103.

(18) Wu, J.; Hua, C.; Liu, B.; Cui, Y.; Zhu, Q.; Xiao, G.; Wu, S.; Cao, G.; Lu, Y.; Ren, Z. Doping-induced superconductivity in the topological semimetal Mo<sub>5</sub>Si<sub>3</sub>. *Chem. Mater.* **2020**, *32*, 8930–8937.

(19) Jorda, J. L. In Mo based alloys and compounds (continued): Datasheet from Landolt-Börnstein - Group III Condensed Matter; "Ac-Na" in SpringerMaterials 21A; Flükiger, R., Klose, W., Eds.; Springer-Verlag Berlin Heidelberg: 1990.

(20) Padamsee, H.; Neighbor, J.; Shiffman, C. Quasiparticle phenomenology for thermodynamics of strong-coupling superconductors. J. Low Temp. Phys. **1973**, *12*, 387–411.

(21) McMillan, W. L. Transition Temperature of Strong-Coupled Superconductors. *Phys. Rev.* **1968**, *167*, 331.

(22) Ito, K.; Hayashi, T.; Nakamura, H. Electrical and thermal properties of single crystalline  $Mo_5X_3$  (X= Si, B, C) and related transition metal 5–3 silicides. *Intermetallics* **2004**, *12*, 443–450.

(23) Jorda, J.; Flükiger, R.; Muller, J. On the formation and stability of the A15 phase in the niobium-gallium system. *J. Less-Common Met.* **1977**, 55, 249–264.

(24) Shishido, T.; Ukei, K.; Toyota, N.; Sasaki, T.; Watanabe, Y.; Motai, K.; Fukuda, T.; Takeya, H.; Takei, H. Flux growth of a new ternary superconducting crystal Nb<sub>5</sub>Sn<sub>2</sub>Ga. *J. Cryst. Growth* **1989**, *96*, 1–6.

(25) Shishido, T.; Ye, J.; Toyota, N.; Ukei, K.; Sasaki, T.; Horiuchi, H.; Fukuda, T. Growth and superconductivity of a new ternary intermetallic compound, Ta<sub>5</sub>Ga<sub>2</sub>Sn. *Jpn. J. Appl. Phys., Part 1* **1989**, 28, 1519.

(26) Claeson, T.; Ivarsson, J.; Rasmussen, S. Superconductivity of Nb<sub>5</sub>Ge<sub>3</sub>. *J. Appl. Phys.* **1977**, *48*, 3998.

(27) Xie, W.; Luo, H.; Phelan, B.; Cava, R.  $Zr_5Sb_{3-x}Ru_x$ , a new superconductor in the  $W_5Si_3$  structure type. *J. Mater. Chem. C* 2015, *3*, 8235–8240.

(28) Xie, W.; Luo, H.; Seibel, E. M.; Nielsen, M. B.; Cava, R. J. Superconductivity in  $Hf_5Sb_{3-x}Ru_x$ : Are Ru and Sb a Critical Charge-Transfer Pair for Superconductivity? *Chem. Mater.* **2015**, 27, 4511–4514.

(29) Wu, J.; Liu, B.; Cui, Y.; Wang, H.; Wang, Z.; Ren, Z.; Cao, G. Type-II superconductivity in  $W_SSi_3$ -type  $Nb_SSn_2Al$ . Supercond. Sci. Technol. **2019**, 32, 045010.

(30) Kawashima, K.; Muranaka, T.; Kousaka, Y.; Akutagawa, S.; Akimitsu, J. Superconductivity in transition metal-silicide  $W_SSi_3$ . J. Phys.: Conf. Ser. 2009, 150, 052106.

(31) Matthias, B. T. Empirical Relation between Superconductivity and the Number of Valence Electrons per Atom. *Phys. Rev.* **1955**, *97*, 74–76.

### Recommended by ACS

#### Structural Phase Transitions and Superconductivity Induced in Antiperovskite Phosphide CaPd<sub>3</sub>P

Akira Iyo, Kenji Kawashima, et al. AUGUST 26, 2020 INORGANIC CHEMISTRY

pubs.acs.org/IC

#### From Ta<sub>2</sub>S<sub>5</sub> Wires to Ta<sub>2</sub>O<sub>5</sub> and Ta<sub>2</sub>O<sub>5-x</sub>S<sub>x</sub>

Shermane M. Benjamin, John J. Neumeier, et al. FEBRUARY 18, 2021 ACS OMEGA READ

# Superconductivity in the Electron-Doped Chevrel Phase Compound $Mo_6S_{6.8}Te_{1.2}$

Feng Lin, Fuqiang Huang, et al. MAY 05, 2020 INORGANIC CHEMISTRY

READ 🗹

READ 🗹

#### Thermoelectric Properties of the As/P-Based Zintl Compounds $EuIn_2As_{2-x}P_x$ (x = 0-2) and $SrSn_2As_2$

Keisuke Shinozaki, Yoshikazu Mizuguchi, et al.

APRIL 22, 2021	
ACS APPLIED ENERGY MATERIALS	READ 🗹

Get More Suggestions >