



Strong-coupling superconductivity with $T_c \sim 10.8$ K induced by P doping in the topological semimetal Mo_5Si_3

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ABSTRACT By performing P doping on the Si sites in the topological semimetal Mo_5Si_3 , we discover strong-coupling superconductivity in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0.5 \leq x \leq 2.0$). Mo_5Si_3 crystallizes in the W_5Si_3 -type structure with space group of $I4/mcm$ (No. 140), and is not a superconductor itself. Upon P doping, the lattice parameter a decreases while c increases monotonously. Bulk superconductivity is revealed in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0.5 \leq x \leq 2.0$) from resistivity, magnetization, and heat capacity measurements. T_c in $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ reaches as high as 10.8 K, setting a new record among the W_5Si_3 -type superconductors. The upper and lower critical fields for $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ are 14.56 T and 105 mT, respectively. Moreover, $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ is found to be a fully gapped superconductor with strong electron-phonon coupling. First-principles calculations suggest that the enhancement of electron-phonon coupling is possibly due to the shift of the Fermi level, which is induced by electron doping. The calculations also reveal the nontrivial band topology in Mo_5Si_3 . The T_c and upper critical field in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ are fairly high among pseudobinary compounds. Both of them are higher than those in NbTi, making future applications promising. Our results suggest that the W_5Si_3 -type compounds are ideal platforms to search for new superconductors. By examinations of their band topologies, more candidates for topological superconductors can be expected in this structural family.

Keywords: Mo_5Si_3 , superconductivity, doping, topological insulator, electron-phonon coupling

INTRODUCTION

Topological superconductors, hosting both gapped bulk superconducting states and gapless surface states, have attracted much attention in recent years. The most fascinating feature in a topological superconductor is that its quasiparticle excitations form Majorana zero modes (MZMs) at boundaries and vortices [1–3]. The MZMs obey the non-Abelian statistics, and are thus

suitable for the realization of fault-tolerant quantum computations [4,5].

In view of the above intriguing merits, much effort has been devoted to seeking for topological superconductivity in real materials. One approach is to search for superconductors with odd parity, as demonstrated in Sr_2RuO_4 , $\text{Sn}_{1-x}\text{In}_x\text{Te}$, or Td-MoTe_2 [6–9]. However, odd-parity superconductors are very rare, and their superconductivity is generally fragile to impurities or disorders. Moreover, all these superconductors have very low (< 2 K) superconducting transition temperatures (T_c), limiting possible applications. Another approach, as proposed by Fu and Kane [10], is to fabricate heterostructures made of superconductors and topological insulators, where the topological surface states (TSSs) become superconducting from the proximity effect. The realizations [11–13], however, require delicate device fabrications and face the challenges of lattice mismatch and interface complexity.

Consequently, researchers in this field have been moving much of their attention to find bulk superconductors that also possess nontrivial band topologies. Such conception is simple but effective. In this spirit, superconducting topological materials such as $\beta\text{-PdBi}_2$, doped Bi_2Se_3 , PdBi , and 2M-WS_2 were proposed to be candidates for topological superconductors [14–21]. In many of them, spectroscopy methods such as scanning tunneling microscopy (STM) and angle-resolved photoemission spectroscopy (ARPES) successfully confirmed the existence of TSSs [20–23], which in turn verified the effectiveness of the conception.

Recently, superconductivity was observed in Re-doped Mo_5Si_3 , with a maximal T_c of 5.8 K in $\text{Mo}_3\text{Re}_2\text{Si}_3$ [24]. Not only did Ref. [24] set a record-high T_c in W_5Si_3 -type superconductors, it also emphasized the nontrivial band topology, making $\text{Mo}_3\text{Re}_2\text{Si}_3$ a candidate for topological superconductors. We noticed that, just like the cases of Cu/Sr/Nb-doped Bi_2Se_3 [14–17,19], superconductivity was successfully induced by carrier doping in topological material Mo_5Si_3 . We thus systematically examined the doping effects not only on the Mo sites, but also on the Si

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sites in Mo_5Si_3 .

In this paper, we report detailed characterizations of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$), in which bulk superconductivity with T_c as high as 10.8 K is observed. In addition, $\text{Mo}_5\text{Si}_2\text{P}$ and $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ are found to host strong electron-phonon coupling. The enhancement of the coupling strength is due to the shift of the Fermi level, and possibly the phonon softening, as revealed by the heat capacity measurements and first-principles calculations. A series of superconducting parameters for $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ are determined, and the electronic band topologies are briefly discussed.

EXPERIMENTAL SECTION

Polycrystalline samples of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($x = 0, 0.5, 1.0, 1.2, 1.3, 1.5, 1.6$, and 2.0) were prepared by solid state reaction. Elements of Mo (99.9%, powder), Si (99.999%, powder), and P (99.99%, powder) were mixed thoroughly before being pressed into pellets. The pellets were placed into alumina crucibles before being sealed into silica tubes under argon. The tubes were slowly heated to 1073 K and held for 24 h. Then the products were thoroughly ground, pressed into pellets, put into alumina crucibles, and sealed into tantalum tubes under argon. The tubes were heated under high-purity argon at 1923 K for 20 h. All the manipulations except sealing and heating were carried out in a glove box filled with high-purity argon. The final products showed silver metallic lusters and were stable in air.

The room-temperature powder X-ray diffraction (XRD) data were collected on a PAN-analytical X-ray diffractometer with Cu-K α radiation. Rietveld refinements were carried out using the GSAS package [25]. The resistivity and heat capacity data were collected on a physical property measurement system (PPMS, Quantum Design). The magnetization measurements were performed on a magnetic property measurement system (MPMS, Quantum Design). The chemical compositions were determined by an energy-dispersive X-ray (EDX) spectrometer equipped on a Phenom ProX scanning electron microscope. More details about the measurements can be found in our previous study [26].

First-principles calculations were performed based on the density functional theory (DFT), as implemented in the Quantum ESPRESSO package [27]. The exchange-correlation functionals of Perdew-Burke-Ernzerhof (PBE) based on the generalized gradient approximation (GGA) were chosen. The optimized norm-conserving pseudopotentials [28] were used. Before the calculations for charge densities, the lattice parameters, as well as the atomic positions, were fully relaxed until the force on each atom was less than $0.0001 \text{ Ry Bohr}^{-1}$. A Monkhorst-Pack grid of $15 \times 15 \times 11$ was applied in the self-consistent calculations. P doping on the Si sites was treated by the method of virtual crystal approximation (VCA). The validation of VCA has been checked with the supercell results (see Fig. S1).

RESULTS

Structural characterizations

Fig. 1a demonstrates the crystal structure of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$, where one may notice that the structure features Si-Si chains along the c axis. XRD patterns of polycrystalline $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$) are shown in Fig. 1e. Without P doping, a phase pure Mo_5Si_3 sample, which is of the tetragonal W_5Si_3 type (space group $I4/mcm$), is successfully obtained. Upon doping, an impurity phase of Mo_3P emerges. Using MoP precursor instead of P in the preparation procedure was found to be beneficial to reducing the amount of Mo_3P in the final products. However, we were not able to completely remove the Mo_3P impurity. This is possibly due to the inevitable evaporation of P at high temperatures. As a result, the actual contents of P in the products should be less than the nominal ones, which was confirmed by our EDX measurements (Fig. S2). The measured values of x are listed in Table 1, and are plotted in Fig. 1d. Note that the measured P contents are not far from the nominal ones. For simplification, x in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ mentioned hereafter represents the nominal value.

The diffraction peaks of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ evidently shift with increasing x , as shown in Fig. 1f. To gain insights into the crystallographic parameters, we performed Rietveld refinements to each XRD pattern. Details of the refinement results are listed in Table 1. Two of them ($x = 0.5$ and $x = 1.5$) are shown as examples in Fig. 1b. The small values of R_p , R_{wp} , and χ^2 suggest the refinements are satisfactory. As shown in Fig. 1c, the a -axis shrinks while the c -axis expands monotonously, indicating a successful P doping into Mo_5Si_3 . This doping behavior is different from that in $\text{Mo}_{5-x}\text{Re}_x\text{Si}_3$, where only the a -axis was changed [24]. We note that the shrinkage of a -axis of Mo_5SiP_2 compared with Mo_5Si_3 is around 1.6%, while the changes of the lattice parameters of Mo_3P impurity are less than 0.1%. This means that the Si doping content in Mo_3P is insignificant (if not zero) in our samples. There are two different Wyckoff positions (Si1 at $4a$ and Si2 at $8h$) of Si in Mo_5Si_3 . Therefore, there could be a site-selection in P doping. We carefully examined the evolution of all the bond lengths in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$. However, no evidence backing this assumption was found. No reflections from the supercell were observed in the XRD patterns either. P is thus believed to randomly take all the Si sites (we should note that it is generally very difficult to distinguish P from Si by XRD measurements, so chances are that there still exists nonequivalent doping between the Si1 and Si2 sites). The shrinkage of a and expansion of c are consistent with our DFT relaxation results (see Fig. S3).

Superconducting properties

The temperature dependence of electrical resistivity (ρ) for $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$) is shown in Fig. 2a. All of the samples show metallic behaviors. For the undoped sample Mo_5Si_3 , ρ reads $\sim 0.22 \text{ m}\Omega \text{ cm}$ at 300 K and decreases monotonously with the decrease of temperature. $\rho(T)$ for Mo_5Si_3 shows no superconducting transitions down to 1.8 K. These results are in good agreement with those in the literature [24,29], where no superconductivity was observed above 0.15 K. P doping into Mo_5Si_3 introduces superconductivity, as revealed by abrupt drops of $\rho(T)$ curves for the doped samples. The region of the superconducting transitions is emphasized in Fig. 2b. For the samples with $x \geq 1.0$, the normal state $\rho(T)$ curves obviously show upwards concave features, similar to those observed in the A15 compounds, which can be interpreted by a parallel-resistor model [30].

Superconducting properties

To investigate the magnetic properties of the superconducting samples, the direct-current (DC) magnetic susceptibility ($4\pi\chi$) of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0.5 \leq x \leq 2.0$) was measured and is shown in Fig. 2c. Note that the data have been corrected with the corresponding demagnetization factors. In the zero-field-cooling (ZFC) run, $4\pi\chi$ of each sample quickly approaches a constant at low tem-

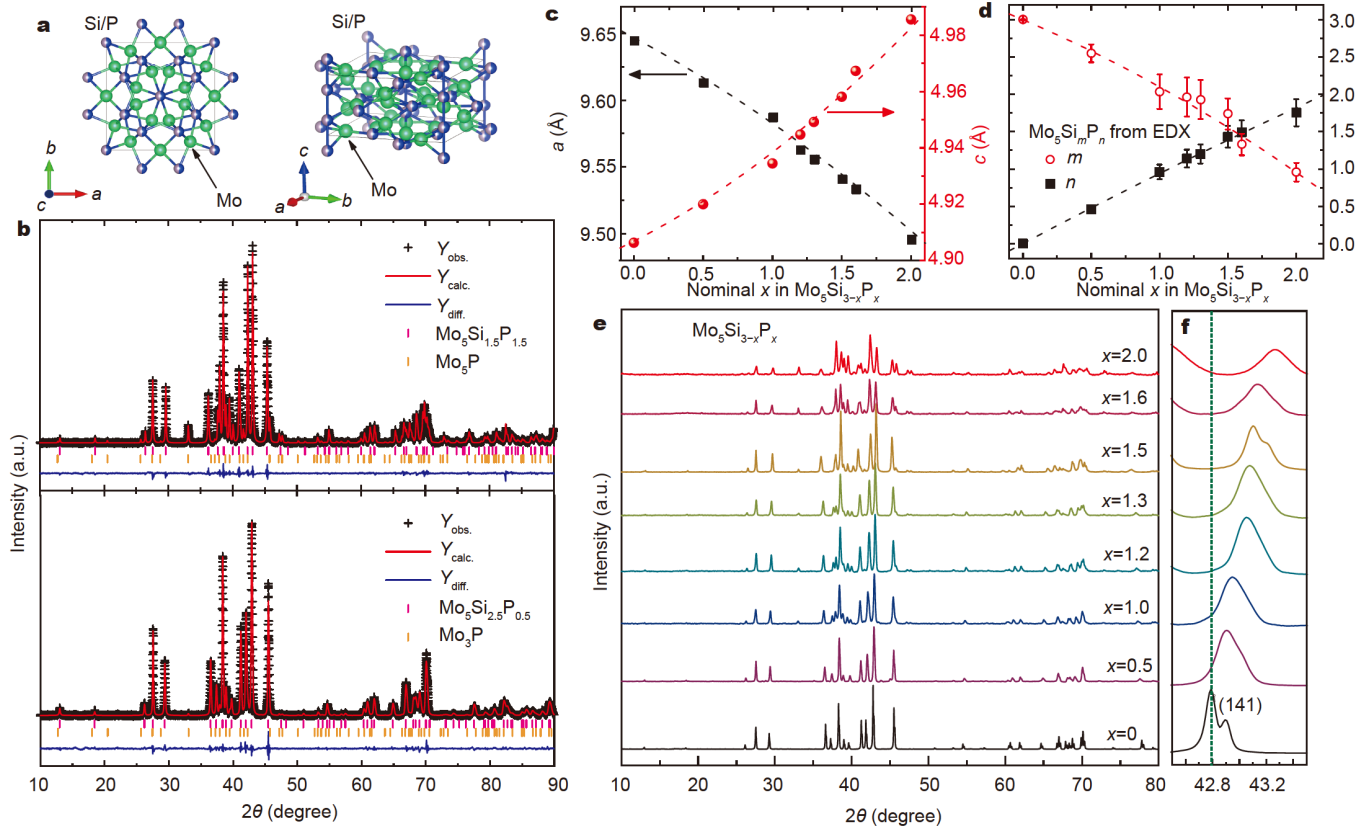


Figure 1 (a) Crystal structure of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$. (b) XRD patterns of $\text{Mo}_5\text{Si}_{2.5}\text{P}_{0.5}$ and $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ with their Rietveld refinements. (c) Evolution of lattice parameters a and c upon P doping. (d) The composition $\text{Mo}_5\text{Si}_m\text{P}_n$ determined from EDX measurements. (e) XRD patterns of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$), with a zoom-in of the (141) peak shown in (f).

Table 1 Crystallographic parameters, measured P contents (EDX), and superconducting T_c of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$)

parameter	$x = 0$	$x = 0.5$	$x = 1.0$	$x = 1.2$	$x = 1.3$	$x = 1.5$	$x = 1.6$	$x = 2.0$
Measured P content x	0	0.46(5)	0.96(9)	1.14(12)	1.20(13)	1.43(14)	1.49(16)	1.75(18)
a (Å)	9.6444(1)	9.6128(1)	9.5871(1)	9.5627(1)	9.5555(1)	9.5407(1)	9.5332(1)	9.4956(2)
c (Å)	4.9063(1)	4.9200(1)	4.9345(1)	4.9447(1)	4.9492(1)	4.9582(1)	4.9674(1)	4.9856(2)
$x_{\text{Si}2}^a$	0.1680(2)	0.1657(3)	0.1656(3)	0.1663(3)	0.1662(2)	0.1666(1)	0.1655(4)	0.1650(4)
$x_{\text{Mo}2}$	0.07694(6)	0.07684(8)	0.0767(1)	0.0771(1)	0.0770(1)	0.07630(4)	0.0767(1)	0.0770(1)
$y_{\text{Mo}2}$	0.22356(7)	0.2232(1)	0.2226(1)	0.2221(1)	0.2219(1)	0.22114(5)	0.2218(2)	0.2214(2)
R_p	1.38%	2.10%	2.11%	2.15%	1.96%	0.94%	1.94%	2.07%
R_{wp}	2.32%	2.89%	2.86%	2.92%	2.62%	1.34%	2.52%	2.75%
χ^2	2.48	2.51	2.08	2.24	1.74	1.56	1.76	1.74
Mo_3P weight fraction ^b	0	3.4%	9.9%	9.4%	6.0%	12.3%	20.6%	30.2%
T_c from $\rho(T)$ (K) ^c	–	6.77	8.70	10.09	10.42	10.74	10.74	10.70
T_c from $\chi(T)$ (K)	–	7.01	9.26	10.18	10.40	10.71	10.54	10.71

a) Wyckoff positions: Si1 (4a), Si2 (8h), Mo1 (4b), Mo2 (16k). P randomly takes the Si1 and Si2 sites. $x_{\text{Si}1} = y_{\text{Si}1} = z_{\text{Si}1} = x_{\text{Si}2} = x_{\text{Mo}1} = z_{\text{Mo}2} = 0$. $y_{\text{Si}2} = x_{\text{Si}2} + 0.5$. $y_{\text{Mo}1} = 2z_{\text{Mo}1} = 0.5$. b) Determined from XRD refinements. c) Determined from the midpoint of superconducting transition in $\rho(T)$.

peratures, indicating the occurrence of superconductivity. The shielding fractions are close to or larger than 100%, validating bulk superconductivity in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$. T_c can be determined from the onset temperature to deviate from the normal states, which are in good agreement with those obtained from the $\rho(T)$ data. It should be mentioned that the diamagnetic signals of Mo_3P ($T_c = 5.6$ K [31]) in the $4\pi\chi(T)$ curves are negligible (it is obvious only in the heavily doped sample Mo_5SiP_2). This is

presumably due to the shielding effects of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$, which have higher T_c than Mo_3P . The absolute values of $4\pi\chi$ in the FC runs are significantly lower than those in the ZFC runs, indicating large pinning effects in the superconducting samples.

Gathering the data from $\rho(T)$ and $4\pi\chi(T)$, we are able to conclude the evolution of T_c in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$), as listed in Table 1, and shown in Fig. 2d. Non-superconducting Mo_5Si_3 becomes superconducting with P doping. T_c quickly increases,

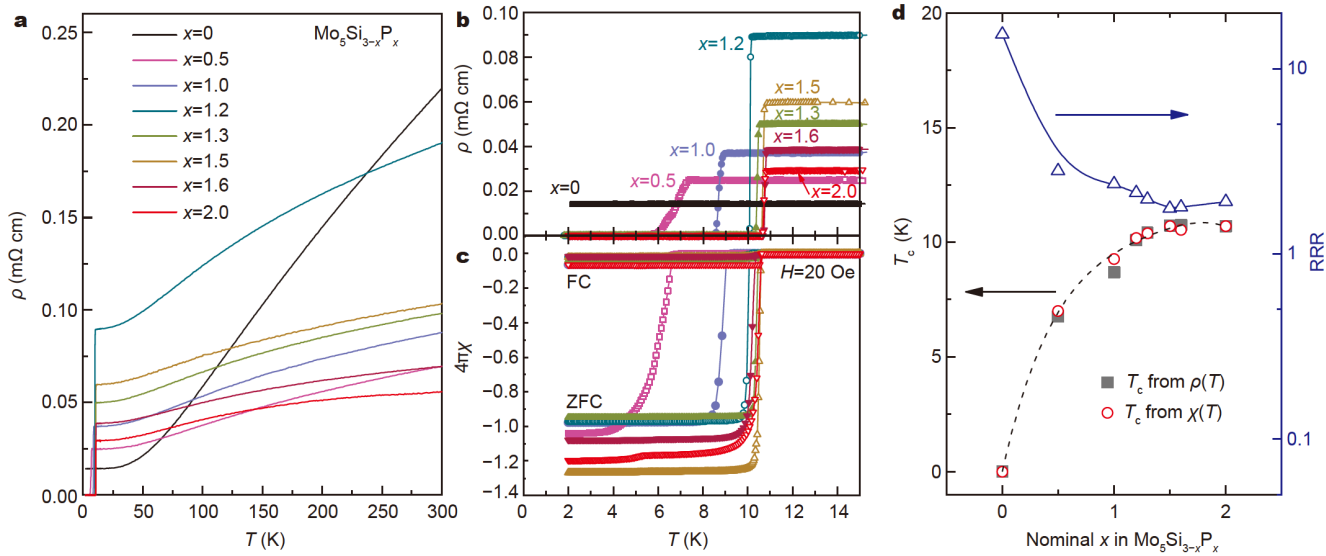


Figure 2 (a) Temperature dependence of resistivity of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$) under zero magnetic field. (b) Zoom-in of the datasets in (a) below 15 K. (c) DC magnetic susceptibility of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0 \leq x \leq 2.0$). (d) Evolution of T_c , as well as RRR, with regard to the P doping content x .

exceeding 10 K in $\text{Mo}_5\text{Si}_{1.8}\text{P}_{1.2}$, while further P doping brings up little change in T_c . Simultaneously, the residual resistivity ratio (RRR) decreases regularly upon P doping, which is reasonable since P doping introduces more defects in the sample.

Now we move on to the discussion of the superconducting nature by conducting a detailed investigation on the $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ sample. We choose to characterize this sample in detail because it hosts the highest T_c , and has less Mo_3P impurities compared with heavier-doped samples. $\rho(T)$ of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ under various magnetic fields ($\mu_0 H = 0 - 15.5$ T) is shown in Fig. 3a. Under zero magnetic field, $\rho(T)$ starts to drop abruptly at a $T_{\text{c onset}}$ of 10.80 K, and reaches zero at a $T_{\text{c zero}}$ of 10.70 K, resulting in a superconducting transition width of only 0.1 K. Upon the application of magnetic field, superconductivity in $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ is gradually suppressed. T_c under different magnetic fields is determined by the 50% criterion, i.e., the temperature where $\rho(T)$ reaches 50% of that of the normal state. The phase diagram of upper critical field ($\mu_0 H_{c2}$) versus T is therefore plotted in Fig. 3d. It can be seen that the Ginzburg-Landau (G-L) model $\mu_0 H_{c2}(T) = \mu_0 H_{c2}(0)[1 - (T/T_c)^2]/[1 + (T/T_c)^2]$ gives a satisfying fit of the experimental results in the whole temperature range. $\mu_0 H_{c2}(0)$ is fitted to be 14.56 T, which is lower than the Pauli paramagnetic limit (20.0 T).

We also performed isothermal magnetization measurements on $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$. The hysteresis loop at 2 K is shown in Fig. 3c. A typical behavior of a type-II superconductor is observed. The isothermal magnetization curves under various temperatures are shown in Fig. 3b, from which the lower critical field ($\mu_0 H_{c1}$) can be determined from the deviation of the curves from the initial Meissner states. $\mu_0 H_{c1}$ under different temperatures is plotted in Fig. 3d. One can easily fit $\mu_0 H_{c1}(T)$ with the well-known G-L expression: $\mu_0 H_{c1}(T) = \mu_0 H_{c1}(0)(1 - t^2)$, where t is the normalized temperature T/T_c . The fit gives a $\mu_0 H_{c1}(0)$ of 105 mT.

Based on the results of $\mu_0 H_{c1}(0)$ and $\mu_0 H_{c2}(0)$, a series of superconducting parameters can be obtained. The G-L coherence length (ξ_{GL}) of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ is calculated to be 4.75 nm by the relation: $\mu_0 H_{c2}(0) = \Phi_0/(2\pi\xi_{\text{GL}}^2)$, in which Φ_0 stands for the magnetic flux quantum. The superconducting penetration depth

(λ_{GL}) is calculated by $\mu_0 H_{c1}(0) = \Phi_0/(4\pi\lambda_{\text{GL}}^2)(\ln\kappa + 0.5)$, where $\kappa \equiv \lambda_{\text{GL}}/\xi_{\text{GL}}$ is the G-L parameter. Consequently, we obtain $\lambda_{\text{GL}} = 70.5$ nm and $\kappa = 14.8$. The value of κ is far larger than $1/\sqrt{2}$, again suggesting $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ to be a type-II superconductor. The thermodynamic critical field can therefore be determined by $\mu_0 H_c(0) = \mu_0 \sqrt{H_{c1}(0)H_{c2}(0)/\ln\kappa}$ to be 0.75 T. All these superconducting parameters are summarized in Table 2.

In order to take more insight into the superconductivity, as well as the thermodynamic properties of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$, we measured the specific heat (C_p) of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($x = 0, 1.0, 1.5$). As the doped samples contained superconducting Mo_3P , we synthesized a phase-pure Mo_3P sample, whose $C_p(T)$ was measured before subtraction from the raw data of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ (see Fig. S4). This approach is similar to that in Mo_5PB_2 [32]. The corrected data are shown in Fig. 4. No superconducting transitions are observed in Mo_5Si_3 , while clear anomalies are found at 8.46 and 10.62 K for $x = 1.0$ and $x = 1.5$, respectively, evidencing the bulk superconductivity. The values of T_c from C_p measurements correspond well with those from the resistivity and the magnetization measurements. For each sample, the behavior of C_p at the normal state up to 18 K can be well described with the Debye model: $C_p(T) = \gamma T + \beta T^3 + \delta T^5$, in which the three terms stand for the Sommerfeld term, the contributions from harmonic phonons and anharmonic phonons, respectively. The fitted curves are shown in Fig. 4a as the dash lines. γ and β for each sample are listed in Table 2. Note that from $x = 0$ to $x = 1.5$, γ almost doubles, while β becomes an order of magnitude larger. We calculate the Debye temperature (Θ_D) by $\Theta_D = (12\pi^4 N R / 5\beta)^{1/3}$, in which N is the number of atoms per formula unit (f.u.), and R is the ideal gas constant. The results are also listed in Table 2. One may see that P doping greatly reduces the value of Θ_D (from 659 to 314 K), implying substantial softening of the lattice. This is quite surprising since the atomic mass of P is close to that of Si. The softening of phonons in P-doped samples may be related to the emergence of phonon soft modes. Detailed theoretical calculations will help to eluci-

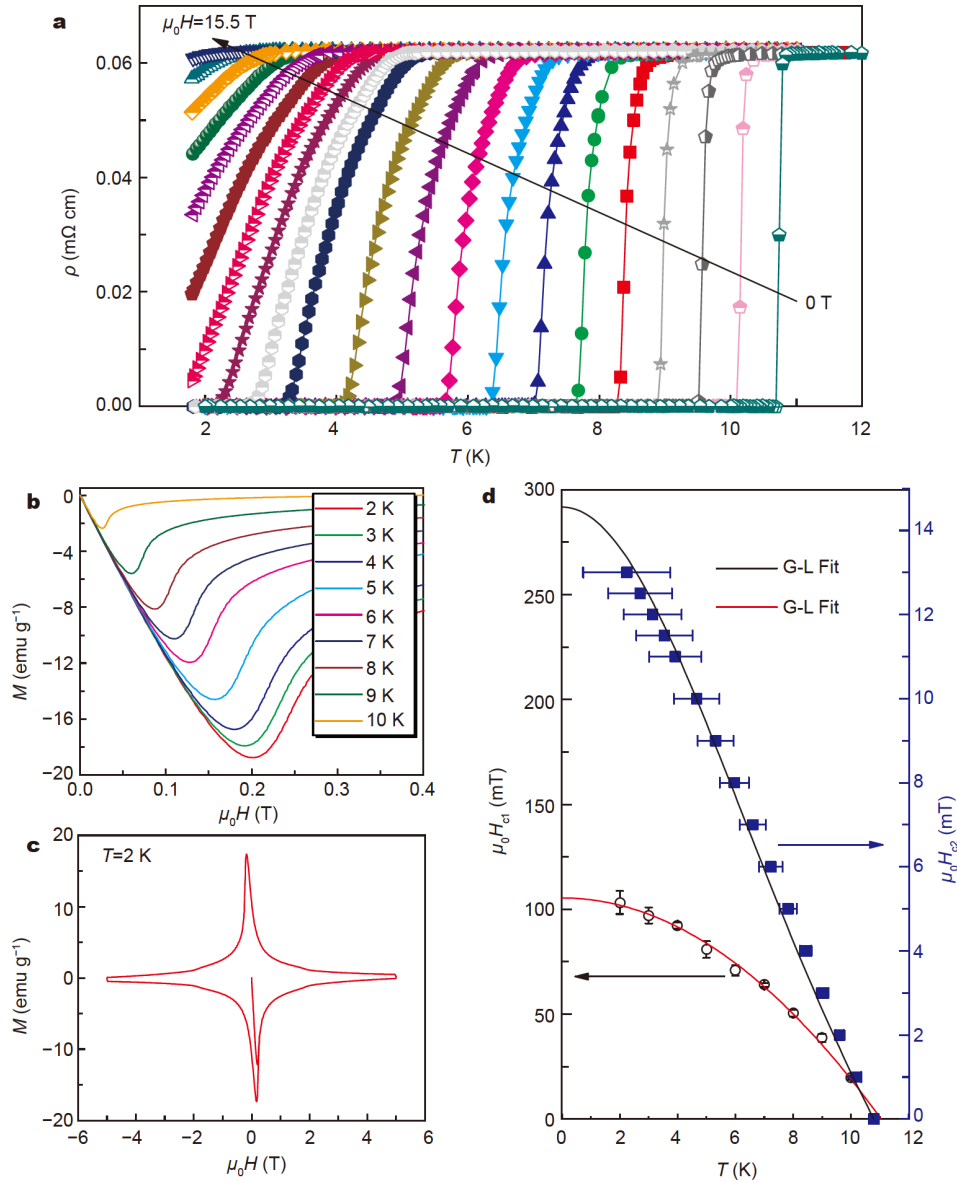


Figure 3 (a) Superconducting transition on $\rho(T)$ of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ under magnetic fields up to 15.5 T. (b) Isothermal magnetization curves of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ at low temperatures. (c) Superconducting hysteresis loop of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ at 2 K. (d) Temperature dependence of the upper and lower critical fields of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$.

date this topic in the future.

The electronic contribution to C_p can thus be obtained by subtracting the phonon terms. Temperature dependence of electronic specific heat (C_e) is shown in Fig. 4b. Note that the normalized C_e jumps at T_c ($\Delta C_e/\gamma T_c$) are 2.03 and 2.06 for $\text{Mo}_5\text{Si}_2\text{P}$ and $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$, respectively. These values are much larger than the Bardeen-Cooper-Schrieffer (BCS) weak coupling ratio (1.43), suggesting strong coupling in these samples.

For a strong-coupling superconductor, the electron-phonon coupling parameter (λ_{ep}) can be estimated by the McMillan formula modified by Allen and Dynes [33,34]:

$$T_c = \frac{\omega_{\ln}}{1.2} \exp \left[\frac{1.04(1 + \lambda_{ep})}{\mu^*(1 + 0.62\lambda_{ep}) - \lambda_{ep}} \right] \quad (1)$$

where the logarithmic average phonon frequency ω_{\ln} is given by [35]:

$$\frac{\Delta C_e}{\gamma T_c} = 1.43 \left[1 + 53 \left(\frac{T_c}{\omega_{\ln}} \right)^2 \ln \left(\frac{\omega_{\ln}}{3T_c} \right) \right] \quad (2)$$

By setting the Coulomb screening parameter $\mu^* = 0.13$, we get $\lambda_{ep} = 1.12$ and 1.15 for $x = 1.0$ and $x = 1.5$, respectively. We further calculate the density of states (DOSs) at the Fermi level using $N(E_F) = 3\gamma/[\pi^2 k_B^2 (1 + \lambda_{ep})]$, where k_B is the Boltzmann constant. The calculations give $N(E_F) = 4.99$ and $7.30 \text{ eV}^{-1} \text{ f.u.}^{-1}$ for $x = 1.0$ and $x = 1.5$, respectively.

C_e in the superconducting state is treated by calculating the entropy with [36]:

$$S(T) = -\frac{3\gamma}{\pi^2 k_B} \int_0^{\pi} \int_0^{\infty} [f \ln f + (1-f) \ln(1-f)] d\epsilon d\varphi, \quad (3)$$

where $f = 1 / \left[1 + \exp \left(\sqrt{\epsilon^2 + \Delta^2(\varphi, T)} / k_B T \right) \right]$ stands for the

Table 2 Superconducting parameters of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$. The thermodynamic parameters of Mo_5Si_3 and $\text{Mo}_5\text{Si}_2\text{P}$ are also listed for comparison.

Parameter	Unit	Mo_5Si_3	$\text{Mo}_5\text{Si}_2\text{P}$	$\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$
$T_{\text{c}}^{\text{onset}}$	K			10.80
$T_{\text{c}}^{\text{zero}}$	K			10.70
$\mu_0 H_{\text{c}1}(0)$	mT			105
$\mu_0 H_{\text{c}2}(0)$	T			14.56
$\mu_0 H_{\text{c}}(0)$	T			0.75
ξ_{GL}	nm			4.75
λ_{GL}	nm			70.5
κ	–			14.8
β	$\text{mJ mol}^{-1} \text{K}^{-4}$	0.054	0.24	0.50
γ	$\text{mJ mol}^{-1} \text{K}^{-2}$	19.80	25.10	37.23
Θ_{D}	K	659	404	314
λ_{ep}	–	–	0.69 ^c	1.15 ^d
$\Delta C_{\text{e}}/\gamma T_{\text{c}}$	–	–	2.03	2.06
$\Delta_0/k_{\text{B}} T_{\text{c}}$	–	–	2.03	2.14
$N(E_{\text{F}})^{\text{a}}$	$\text{eV}^{-1} \text{f.u.}^{-1}$	–	6.30	7.30
$N^{\text{b}}(E_{\text{F}})^{\text{b}}$	$\text{eV}^{-1} \text{f.u.}^{-1}$	4.13	7.24	8.04

a) Experimental value calculated from γ . b) Theoretical value from DFT calculations. c) Estimated from Equation (4). d) Estimated from Equations (1) and (2).

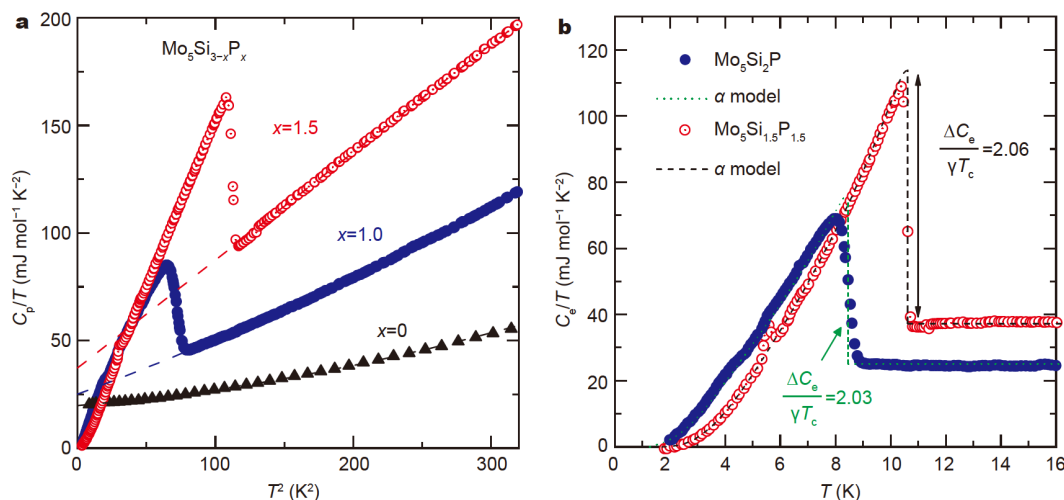


Figure 4 (a) Temperature dependence of heat capacity of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($x = 0, 1.0, 1.5$) under zero magnetic field. The dash lines are fits to the normal state data with the Debye model. (b) Electronic contribution of heat capacity of $\text{Mo}_5\text{Si}_2\text{P}$ and $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$. The data below T_{c} are fitted with the α model, shown as the dash lines. Note that the data for $x = 1.0$ and 1.5 in (a) and (b) have been corrected by subtracting the Mo_3P contributions. The kinks at around 5.6 K are the residual signals from Mo_3P impurity.

Fermi distribution of the quasiparticles. Here, we find that a conventional s -wave gap function reproduces the data well, and the so-called α model is applied. In this model, the angular independent gap function $\Delta(T)$ is expressed as $\Delta(T) = \alpha/\alpha_{\text{BCS}}\Delta_{\text{BCS}}(T)$ where α_{BCS} is the weak-coupling gap ratio (1.76) [37]. C_e is calculated from $C_e = T \frac{\partial S}{\partial T}$. Fittings to the C_e data are illustrated in Fig. 4b, from which we obtain the superconducting gap values at zero temperature $\Delta_0 = 1.48$ and 1.96 meV for $x = 1.0$ and $x = 1.5$, respectively. The coupling strengths $\Delta_0/k_{\text{B}} T_{\text{c}}$ are thus estimated to be 2.03 and 2.14. Again, these values apparently exceed α_{BCS} , evidencing strong-coupled superconductivity.

First-principles calculations

The results of first-principles calculations for $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($x = 0, 1.0, 1.5$) are summarized in Fig. 5, with the electronic band structures of $x = 0$ (with and without spin-orbit coupling (SOC)), $x = 1.0$, and $x = 1.5$ shown in Fig. 5a–d, respectively. For all these samples, there are multiple bands crossing the Fermi level (E_{F}), consistent with the metallic nature of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$. By comparing Fig. 5a, b, we conclude that the SOC has negligible effects on the band structures, although it opens finite gaps on several k points. The shapes of the electronic bands for $\text{Mo}_5\text{Si}_2\text{P}$ and $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ are basically the same with that of Mo_5Si_3 , which means that the bands can be considered rigid in our case. One major feature of the band structure in Fig. 5a is the flat band

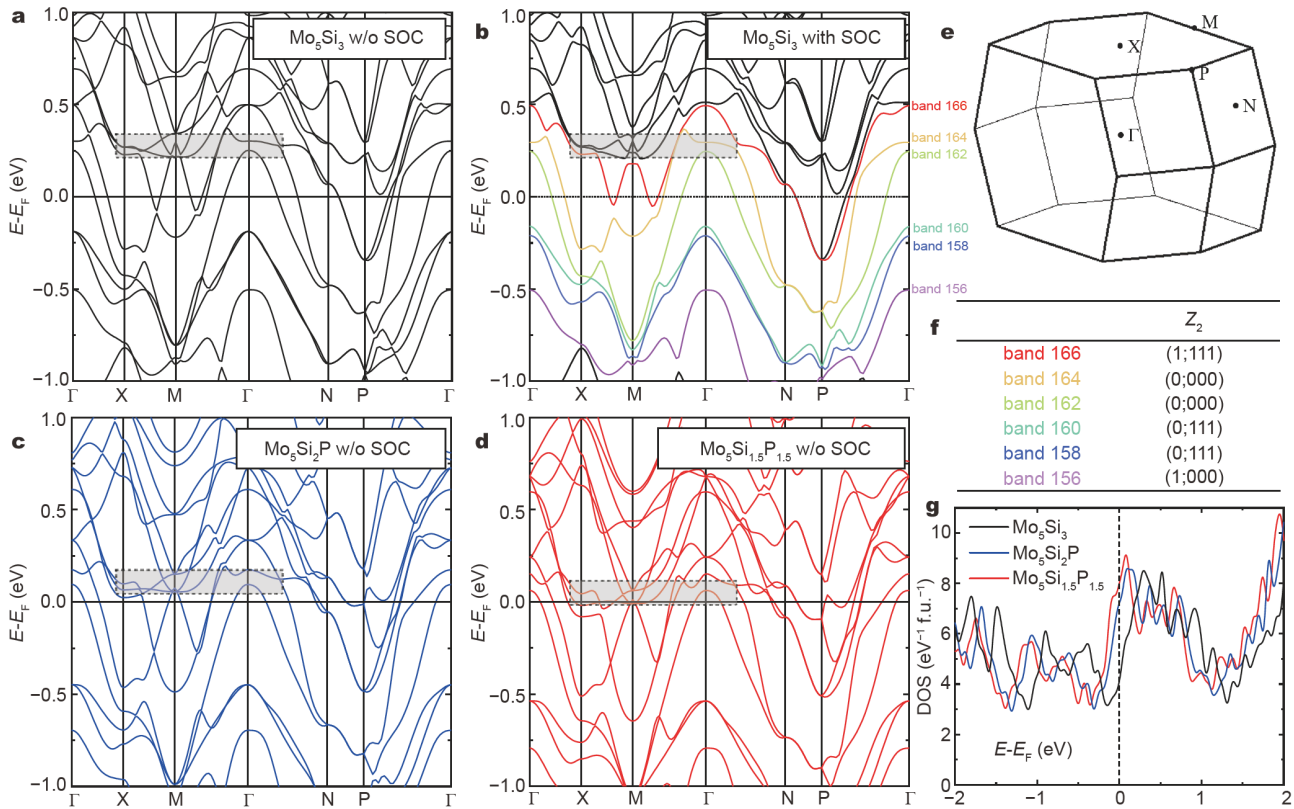


Figure 5 Band structures of (a) Mo_5Si_3 without SOC, (b) Mo_5Si_3 with SOC, (c) $\text{Mo}_5\text{Si}_2\text{P}$ without SOC, and (d) $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ without SOC. The shadowy boxes in (a–d) emphasize the flat band dispersions. (e) Brillouin zone of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$, with high symmetry points labeled. (f) The Z_2 topological invariants of Mo_5Si_3 (with SOC) for the bands near E_F . The band numbers correspond to those in (b). (g) Evolution of DOS of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($x = 0, 1.0, 1.5$) upon P doping.

dispersions at around 0.25 eV above E_F (as indicated by the shadowy box), which gradually shift to E_F when x increases. This process is clearly observed when we examine the evolution of the DOS upon P doping (shown in Fig. 5g). E_F of Mo_5Si_3 locates in a dip of DOS, and it shifts to a major peak for $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$. Theoretical values of DOS at E_F ($N(E_F)$) are 4.13, 7.24, and $8.04 \text{ eV}^{-1} \text{f.u.}^{-1}$ for $x = 0, 1.0$, and $x = 1.5$, respectively. The trend of $N(E_F)$ is consistent with the change of the Sommerfeld parameter γ . We notice that $N(E_F)$ of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ corresponds fairly well with the experimental one, while $N(E_F)$ of $\text{Mo}_5\text{Si}_2\text{P}$ is much larger compared with $N(E_F)$. This means that λ_{ep} of $\text{Mo}_5\text{Si}_2\text{P}$ is probably overestimated, and Equation (3) is not applicable. In fact, if we use the inverted McMillan formula [34]:

$$\lambda_{\text{ep}} = \frac{1.04 + \mu^* \ln(\Theta_D / 1.45 T_c)}{(1 - 0.62 \mu^*) \ln(\Theta_D / 1.45 T_c) - 1.04} \quad (4)$$

λ_{ep} and $N(E_F)$ for $\text{Mo}_5\text{Si}_2\text{P}$ are estimated to be 0.69 and $6.30 \text{ eV}^{-1} \text{f.u.}^{-1}$, respectively.

We further examined the band topology of Mo_5Si_3 . Since Mo_5Si_3 possesses both time-reversal symmetry and inversion symmetry, the Z_2 topological invariants can be easily calculated by checking the wavefunction parities on the eight time-reversal invariant k points [38]. As illustrated in Fig. 5b, we calculate the Z_2 indexes for several bands near E_F . Note that not all these Z_2 indexes are well-defined, since there might be no gap between one band and another. Nevertheless, SOC opens finite gaps between bands 160 and 162, and between bands 164 and 166. Z_2 indexes for these two gaps are (0;111) and (0;000), respectively. Our results are consistent with previous studies [24,39], sug-

gesting that bulk Mo_5Si_3 (with SOC) falls into the weak topological insulator state. Unfortunately, this means that the TSSs in Mo_5Si_3 are fragile, and are unlikely to survive with P doping, which inevitably introduces defects. What makes it worse is that E_F shifts to higher energies in the doped samples, which could push the system into a topologically trivial state. In a word, P doping into Mo_5Si_3 may destroy its TSSs, making it less possible to realize topological superconductivity in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$.

DISCUSSION

In McMillan's formalism [34], the electron-phonon coupling strength is given by $\lambda_{\text{ep}} = [N(E_F) \langle I^2 \rangle] / [M \langle \omega^2 \rangle]$, where M stands for the atomic mass, $\langle I^2 \rangle$ and $\langle \omega^2 \rangle$ are averages of the squared electronic matrix elements on the Fermi surface, and of the squared phonon frequencies, respectively. There are thus at least two approaches to enhance λ_{ep} : one is to increase $N(E_F)$, and the other is to lower $\langle \omega^2 \rangle$ (or, in other words, to soften the lattice). In $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$, $N(E_F)$ is enhanced by electron doping, which shifts E_F to a peak in DOS. The lattice has been effectively softened too, as evidenced by the large decrease of Θ_D . These two factors together give rise to the strong electron-phonon coupling, and should be responsible for the emergence of superconductivity in $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$. The reason why P doping softens the lattice so much (unlike Re doping in $\text{Mo}_{5-x}\text{Re}_x\text{Si}_3$, which did not change Θ_D much [24]) is definitely worth further studies. Theoretical calculations of the phonon dispersions of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ will

be illuminating. In particular, large phonon linewidths or soft modes can be expected.

As for the topological properties, although our results suggest that P doping may not be beneficial to the TSSs, there are other ways to chase for topological superconductivity in this system. We notice that the band gap between bands 156 and 158 is topologically nontrivial with strong topological indexes of (1;000) (see Fig. 5f). Therefore, TSSs are likely to survive and topological superconductivity may be realized, if superconductivity can be induced by hole doping in Mo_5Si_3 . Another strategy is to fabricate $\text{Mo}_5\text{Si}_3/\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ heterojunctions to see whether TSS emerges from the proximity effect. Given the fact that high-quality Mo_5Si_3 single crystals are readily available [29], $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ serves as a suitable platform to observe the possibly existing MZMs, which will be an intriguing topic in future ARPES or STM studies.

Lastly, we should mention that the T_c of 10.8 K and $\mu_0H_{c2}(0)$ of 14.56 T are fairly high for a pseudobinary compound. For example, both of them are slightly higher than those in the commercial superconductor NbTi ($T_c = 9.6$ K, $\mu_0H_{c2}(0) = 14$ T) [40]. Most absorbingly, $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ shares much in common with A15 superconductors such as V_3Si or Nb_3Sn . For instance, they have similar T_c , close coupling strength, and all of them show upwards concave features in normal state $\rho(T)$ [30,41]. Recent theoretical studies suggested nontrivial band topologies in some of the A15 superconductors too [42]. Compared with the well-known A15 superconductor family, the W_5Si_3 -type superconductor family, which $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ belongs to, has not been studied in depth. Currently, the family contains about ten members. Most of them superconduct below 4 K, with the maximal T_c of 5.8 K observed in $\text{Mo}_3\text{Re}_2\text{Si}_3$ [24,43–48]. Our study almost doubles this maximum, making it comparable to those in the A15 family. W_5Si_3 -type compounds are hence a fertile ground to be explored for new superconductors.

CONCLUSIONS

To summarize, we discover that P doping introduces superconductivity in non-superconducting Mo_5Si_3 , which hosts a nontrivial band topology. T_c increases with the doping level, reaching 10.8 K in $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$. $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ is a type-II, fully gapped superconductor with strong electron-phonon coupling, as evidenced by the large values of $\Delta C_e/\gamma T_c$, $\Delta_0/k_B T_c$, and λ_{ep} . $\mu_0H_{c1}(0)$ and $\mu_0H_{c2}(0)$ for $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ are 105 mT and 14.56 T, respectively. According to first-principles calculations, the large electron-phonon coupling is related to the increase of $N(E_F)$. $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ sets a new record of T_c in W_5Si_3 -type superconductors. Compared with the previously reported $\text{Mo}_3\text{Re}_2\text{Si}_3$ superconductor [24], or the widely used commercial superconducting material NbTi [40], the higher T_c and inexpensive raw materials of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ make future applications promising. We point out that the superconducting properties of $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ are very similar with those in the A15 superconductors. Our findings suggest that the T_c levels in W_5Si_3 -type superconductors can be comparable to the A15 superconductors. Novel superconductors with higher T_c values can be anticipated in W_5Si_3 -type structural family. In particular, superconductivity, or even topological superconductivity, could be achieved through carrier doping, whose effectiveness has been testified in our study.

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Author contributions Sun JN conceived the project. Ruan BB and Sun JN synthesized the samples and did most of the measurements; Chen Y, Gu YD and Yang QS assisted in some of the measurements; Ruan BB carried out the theoretical calculations and wrote the paper with supports from Zhou MH, Ma MW and Zhao K; Chen GF, Shan L and Ren ZA reviewed the original manuscript; Ren ZA supervised the project. All authors contributed to the general discussion.

Conflict of interest The authors declare that they have no conflict of interest.

Supplementary information Comparison of VCA and supercell results, SEM image and elemental mapping of $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$, the relaxed lattice parameters from DFT compared with the experimental ones, and the subtraction of Mo_3P contribution from the raw data. Supporting data are available in the online version of the paper.



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拓扑半金属 Mo_5Si_3 中磷掺杂诱导 $T_c \sim 10.8$ K的强耦合超导电性

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摘要 通过对拓扑半金属 Mo_5Si_3 的硅位进行磷掺杂, 我们发现了 $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0.5 \leq x \leq 2.0$)中强耦合的超导电性。 W_5Si_3 结构的 Mo_5Si_3 本身并不具有超导性, 随着磷掺杂的增加, 其晶格常数 a 单调减小, 而 c 单调增加。在 $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ ($0.5 \leq x \leq 2.0$)中, 电阻、磁化率和比热测量揭示了其中的体超导特性。 $\text{Mo}_5\text{Si}_{1.5}\text{P}_{1.5}$ 的超导转变温度(T_c)高达10.8 K, 创造了 W_5Si_3 结构超导体的 T_c 纪录, 其上下临界场分别为14.56 T和105 mT, 且是一个具有强电子-声子耦合的全能隙超导体。第一性原理计算表明, 强的电子-声子耦合可能来自于掺磷所引起的费米面的移动, 同时也揭示了 Mo_5Si_3 非平庸的能带拓扑性质。 $\text{Mo}_5\text{Si}_{3-x}\text{P}_x$ 超导体的 T_c 和上临界场在准二元化合物中相当高, 超过了NbTi超导体, 具有潜在的应用价值。本文的结果表明 W_5Si_3 型结构中可能存在更多的新型超导体, 对该体系的研究将有助于拓扑超导体的发现。